



# Optical Spectroscopy of Graphene and Topological Insulators

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# Outline

## *Graphene*

- Introduction
- Graphene on Cu single crystal surfaces
- Nitrogen doped graphene
- Twisted bilayer graphene

## *Topological insulator nanostructures*

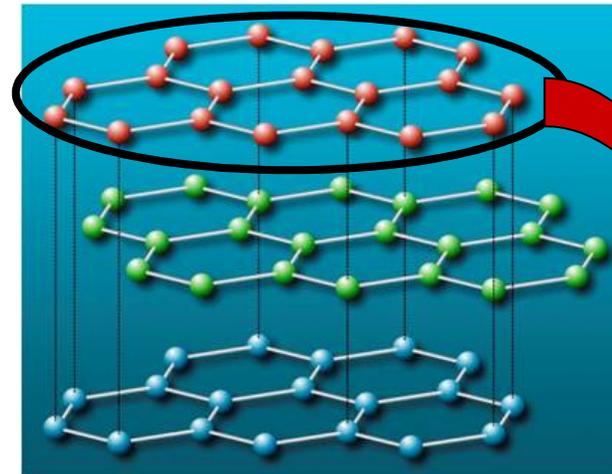
- $\text{Bi}_2\text{Te}_3$  nanoplates

# Graphene

- **Introduction**
- Graphene on Cu single crystal surfaces
- Nitrogen doped graphene
- Twisted bilayer graphene

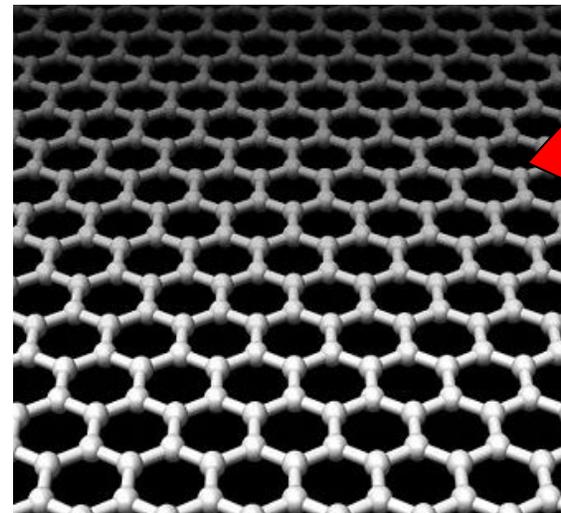
# Graphite → Graphene

Graphite pencils



3D graphite crystal

2D graphene

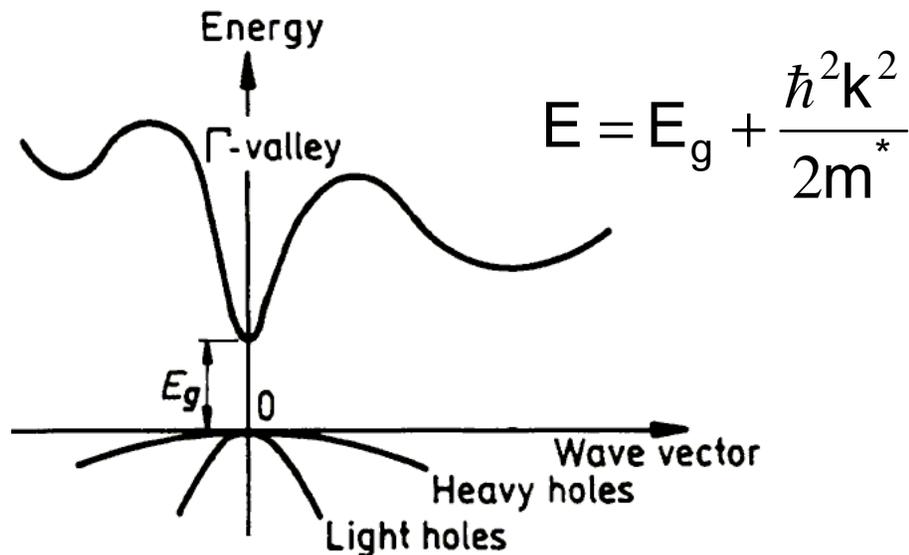


Discovered in  
2004  
Nobel Prize  
awarded in  
2010

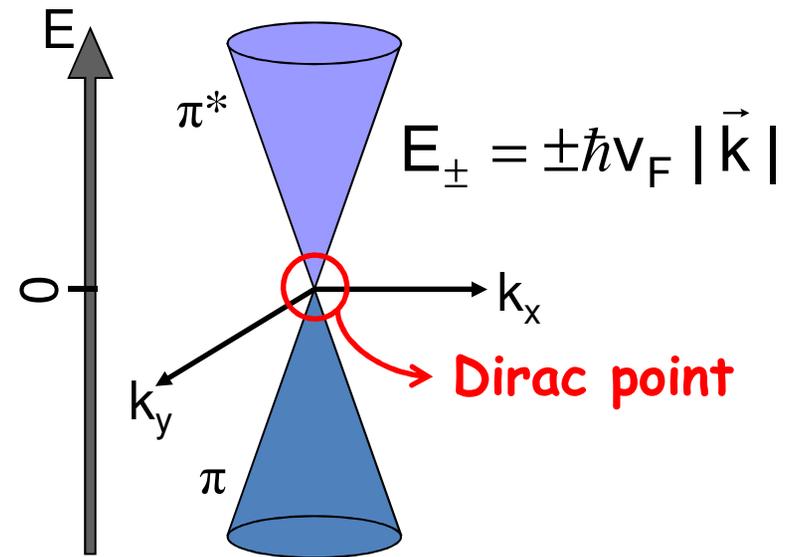
**Graphene** is a **one-atom-thick** ( $\sim 3.35\text{\AA}$ ) planar sheet of carbon atoms that are packed in a **honeycomb** crystal lattice

# Graphene: a new condensed-matter system

*Electron band structure in conventional condensed-matter system*



*Electron band structure in graphene*



Graphene: Linear dispersion near the Dirac points

# Novel properties

- **Low electrical resistivity**  $1\mu\Omega\cdot\text{cm}$  at room temperature  
~33% less than Cu and Ag, the lowest-resistivity metallic materials
- Room temperature **mobility** up to  $200,000\text{cm}^2/\text{Vs}$   
electrons in graphene travel more than 100 times faster than those in Si ( $\mu\sim 1400\text{cm}^2/\text{Vs}$ )
- **Superb mechanical strength**  
breaking strength comparable to that of the diamond
- **Superior thermal conductivity**  
Heat transfer ability ~10 times higher than Cu and Ag
- **Optically transparent in the visible range: transparent conductor**

# Significant potential applications

high-frequency electronics

BREVIA

Science, 327, 662 (2010)

## 100-GHz Transistors from Wafer-Scale Epitaxial Graphene

Y.-M. Lin,\* C. Dimitrakopoulos, K. A. Jenkins, D. B. Farmer, H.-Y. Chiu, A. Grill, Ph. Avouris\*

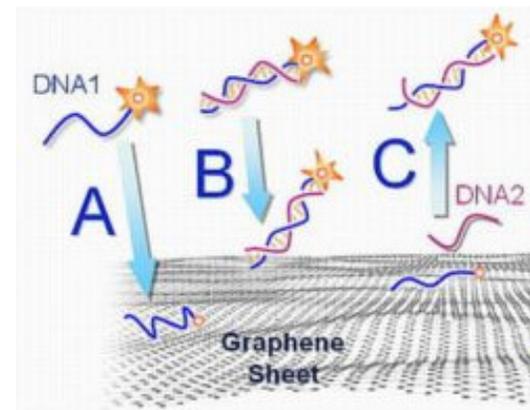
optoelectronic devices  
(displays, touch screens, etc.)



stronger composite materials  
(lighter but more crack-resistant aircraft)



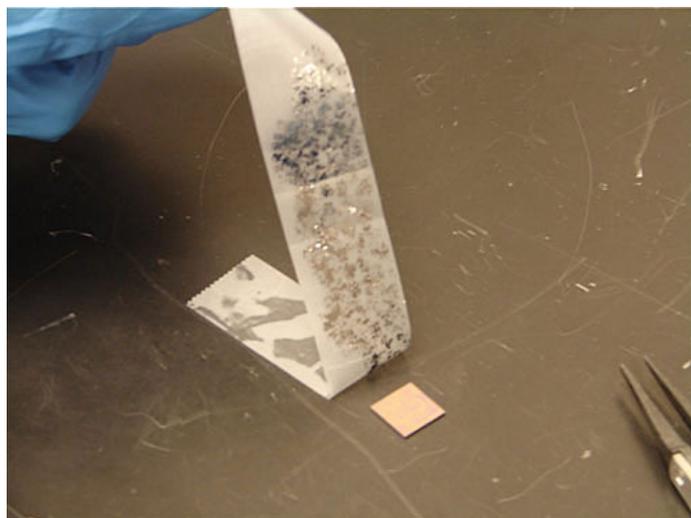
biosensors  
(disease diagnosis)



# How to make graphene?

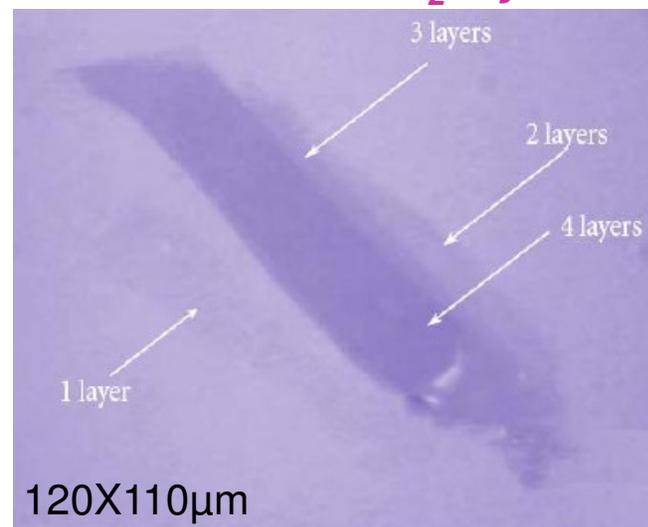
## Scotch-tape method

### Mechanical exfoliation



Scientific American

### Observable color contrast on 300 nm SiO<sub>2</sub> layer



NT-MDT

### ➤ Advantages:

- \* high quality graphene flakes
- \* low cost

### ➤ Drawbacks:

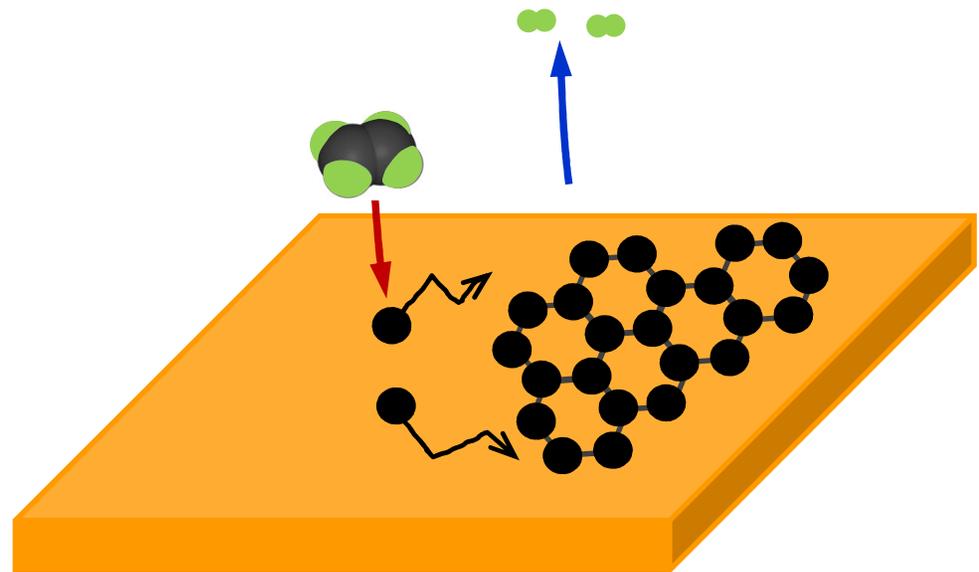
- \* low efficiency (tiny flakes)
- \* not suitable for large area production required by electronic applications

# How to make large area graphene?

## *Growth on transition metal by chemical vapor deposition (CVD)*

- Growth temperature 900-1000°C
- Ethylene or methane is used as C source
- Cu is a catalyst for graphene growth

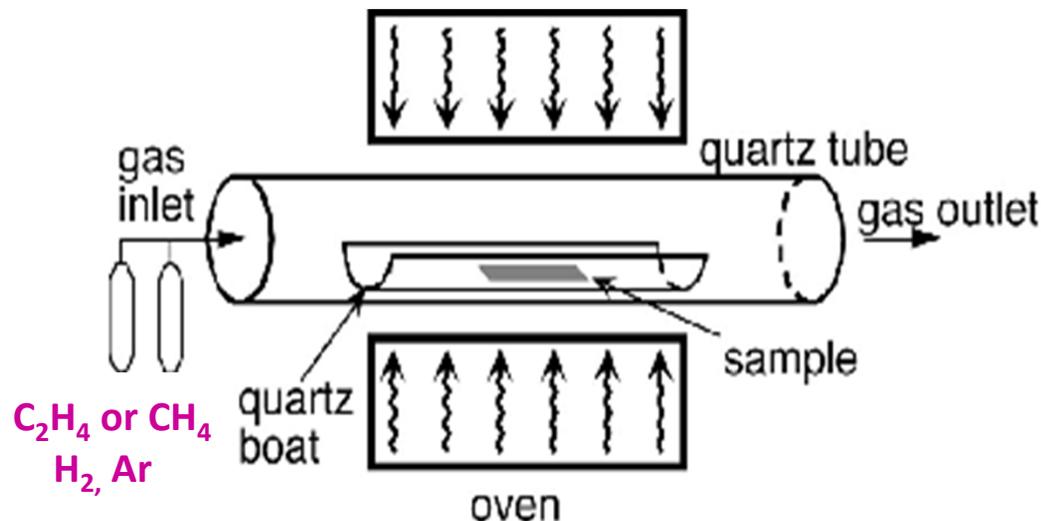
*hydrocarbon molecules  
crack on hot Cu surface*



# How to make large area graphene?

## chemical vapor deposition (CVD)

### apparatus

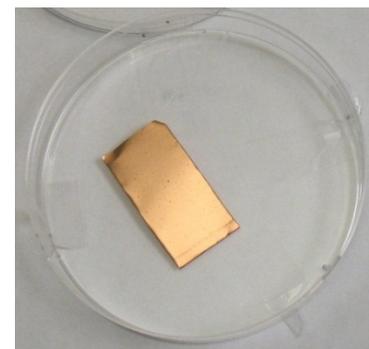


#### ➤ Advantages:

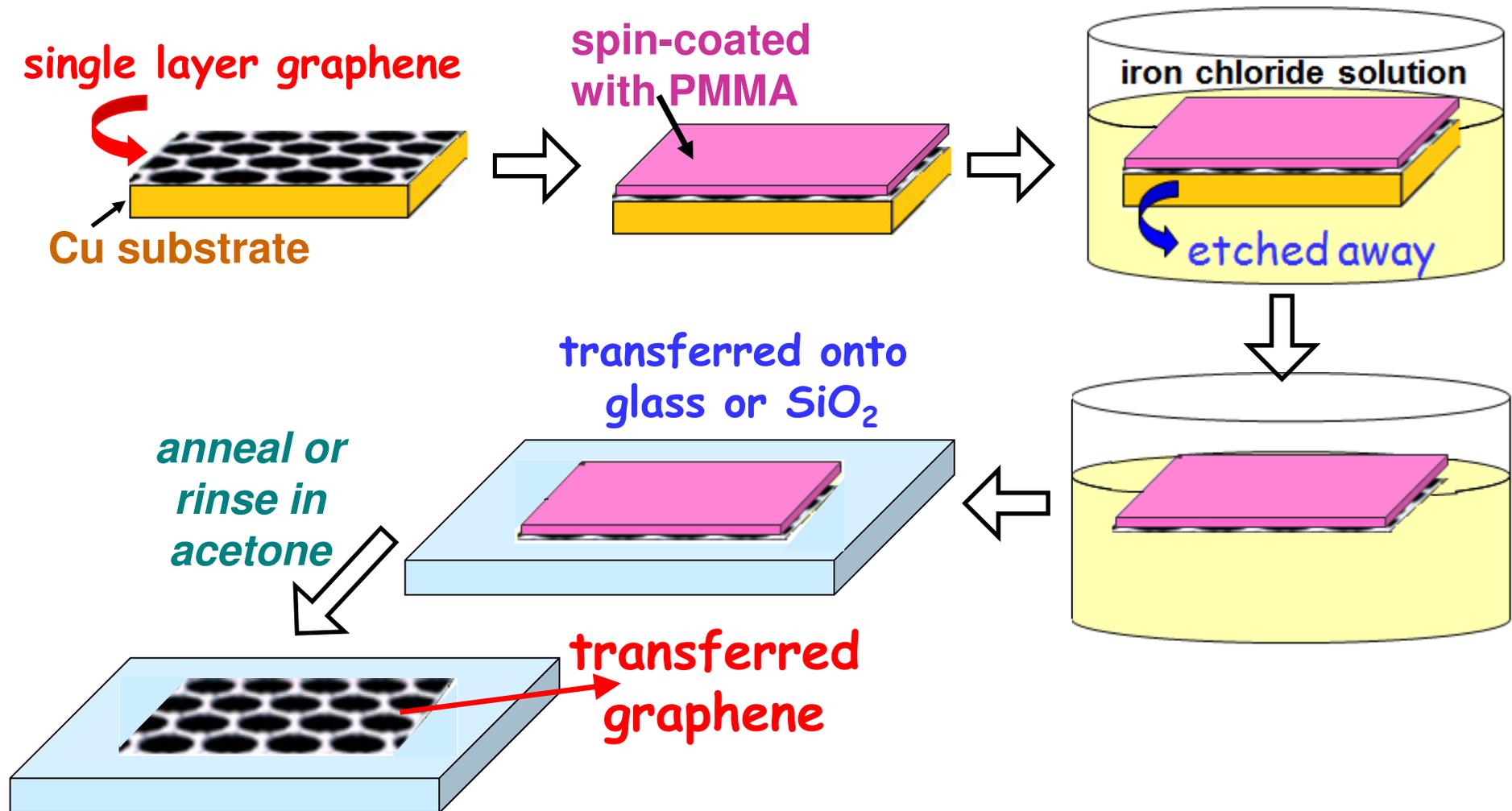
- \* large area graphene
- \* reasonably low cost
- \* ease of processing

#### ➤ Drawbacks:

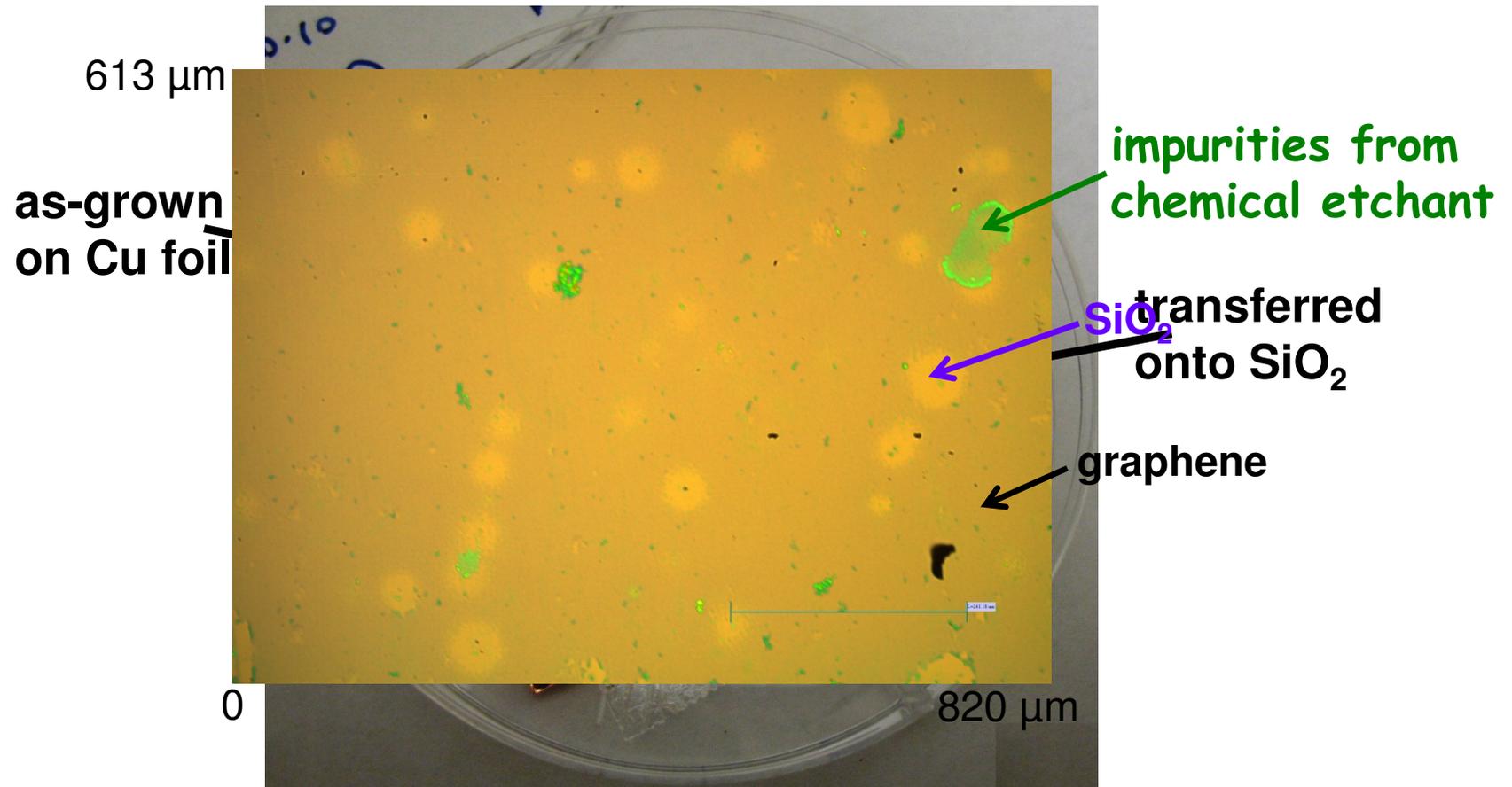
- \* lower quality than exfoliated flakes (natural graphene)
- \* have to **transfer to dielectric substrates** for electronic applications



# Transfer of graphene from metal surfaces onto dielectric substrates



# Graphene after transfer large area

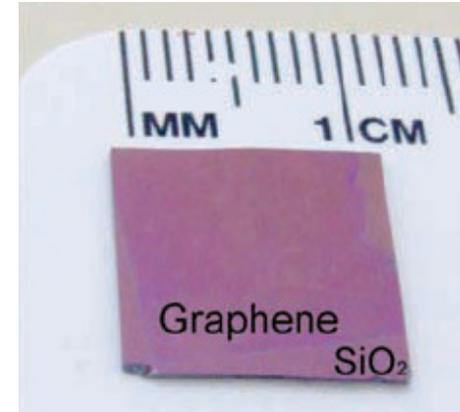


***The transfer process introduces impurities, defects, and mechanical damages in graphene***

# Natural graphene flake has higher quality than CVD graphene



K. Novoselov,  
*et al.*, *Science*,  
**306**, 666 (2004)



X. Li, *et al.*,  
*Science*, **324**,  
1312 (2009)

**Natural graphene flake on  
SiO<sub>2</sub>:**  
 $\mu \sim 10,000\text{-}25,000\text{cm}^2/\text{Vs}$

**CVD graphene on Cu foil  
transferred onto SiO<sub>2</sub>:**  
 $\mu \sim 2000\text{-}5000\text{cm}^2/\text{Vs}$

**We study CVD growth on single crystal Cu**

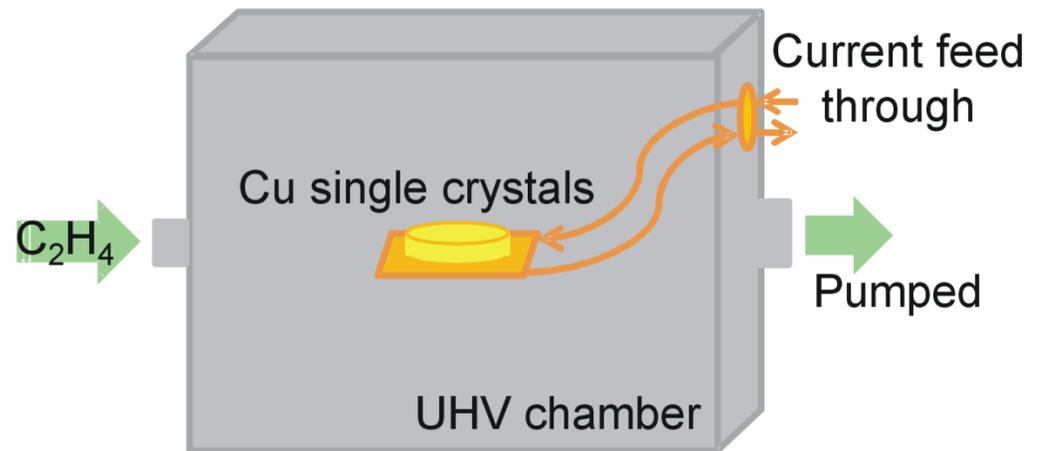
# Graphene

- Introduction
- **Graphene on Cu single crystal surfaces**
- Nitrogen doped graphene
- Twisted bilayer graphene

# Graphene growth on Cu single crystal

*Chemical vapor deposition (CVD)  
in UHV (STM) environment*

- Growth temperature 900°C
- Ethylene was used as C source
- Cu is a catalyst for graphene growth



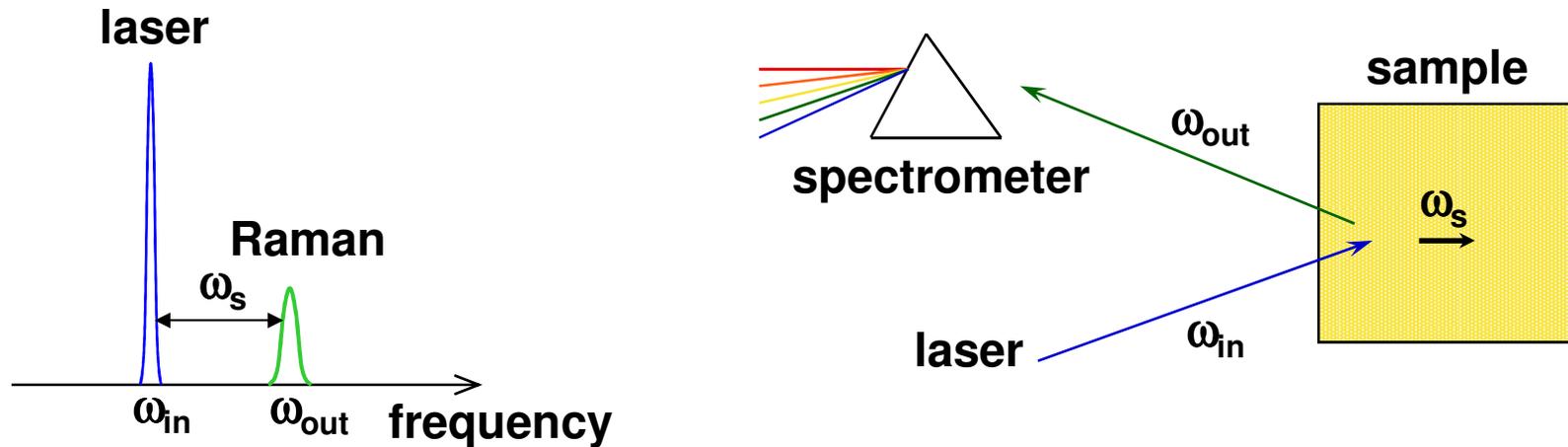
# Graphene on Cu single crystal surfaces

## Graphene layers on Cu surfaces are strained

- strains in graphene on Cu single crystal (111) and (100) surfaces are very different
- strain is released after transfer

The quality of CVD graphene on Cu single crystal is high

# Inelastic light scattering: Raman scattering



scattering frequency

$$\omega_s = \omega_{in} - \omega_{out}$$

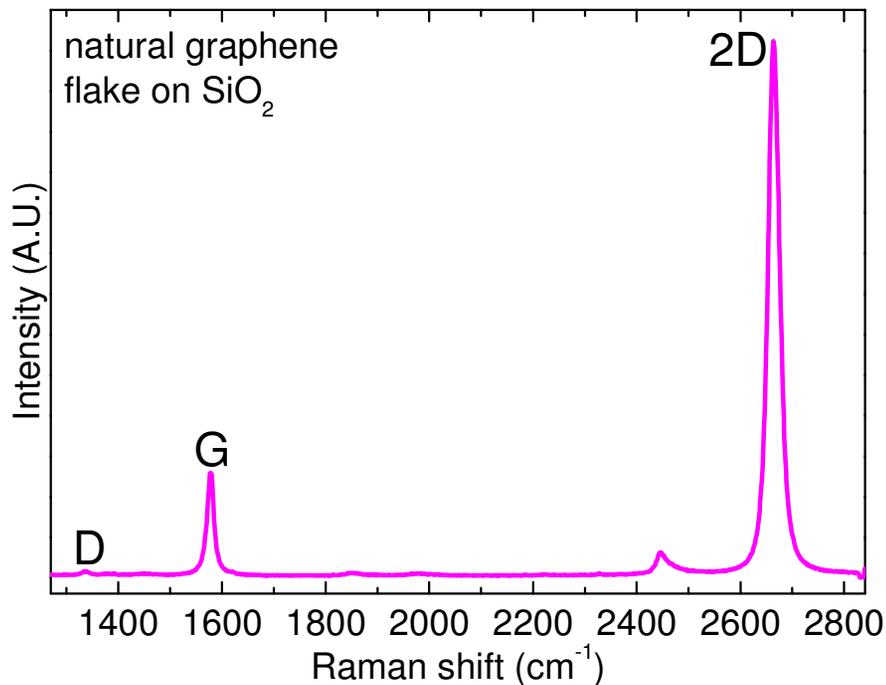
scattering wave vector

$$\vec{k}_s = \vec{k}_{in} - \vec{k}_{out}$$

Raman scattering probes elementary excitations:  
phonons, vibrations, electronic excitations, etc.

# Characterization of graphene by Raman scattering

*Raman spectrum*



## Measure optical phonons

**G-band** ( $q \rightarrow 0$  mode) at  $\sim 1580\text{cm}^{-1}$

(electron-phonon coupling)

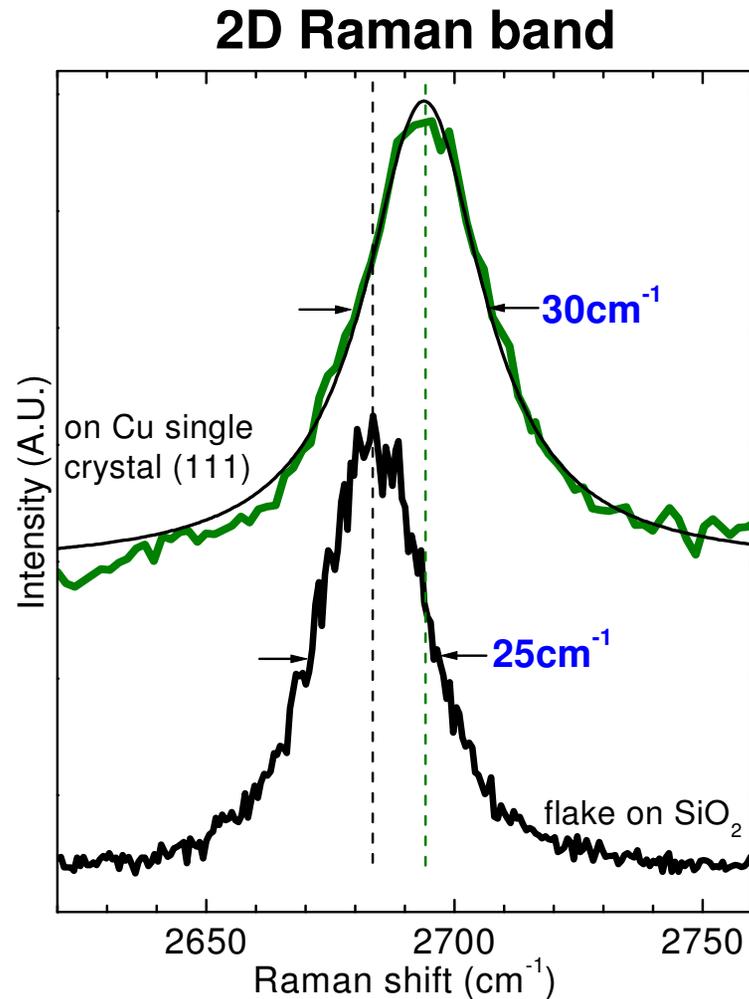
**2D-band** (two-phonon mode) at

$\sim 2680\text{cm}^{-1}$  (monitors strain)

**D-band** at  $\sim 1350\text{cm}^{-1}$  (due to

disorder)

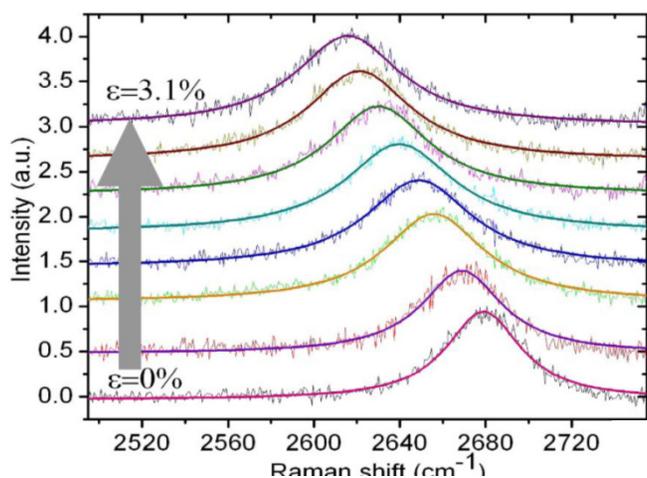
# CVD graphene on (111) Cu single crystal



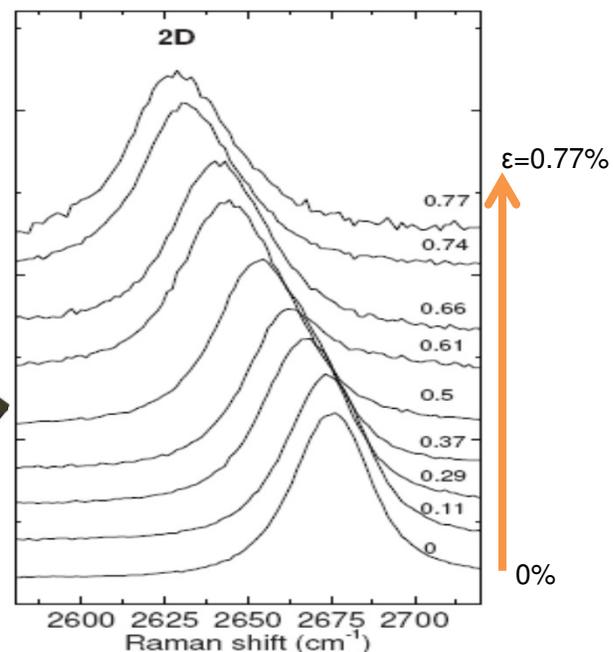
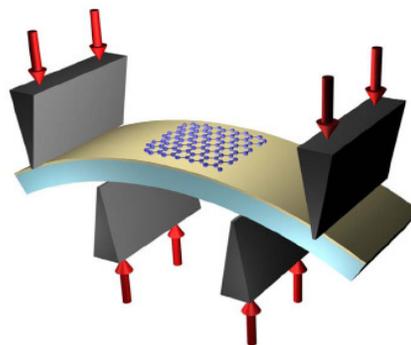
- 2D Raman band is **blue-shifted**
- **quality comparable to natural graphene!** (*similar FWHM*)

# 2D Raman band is very sensitive to strain

*mode softens under tensile strain*



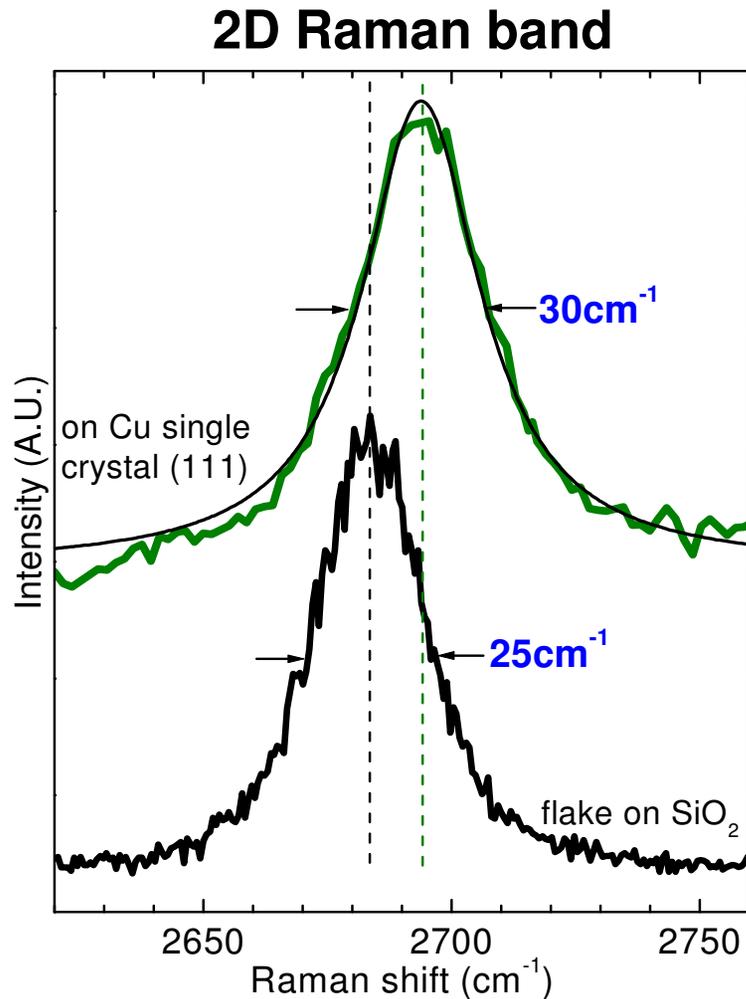
M. Huang, *et al.*, PNAS, **106**, 7304 (2009)



T. Mohiuddin, *et al.*, PRB, **79**, 205433 (2009)

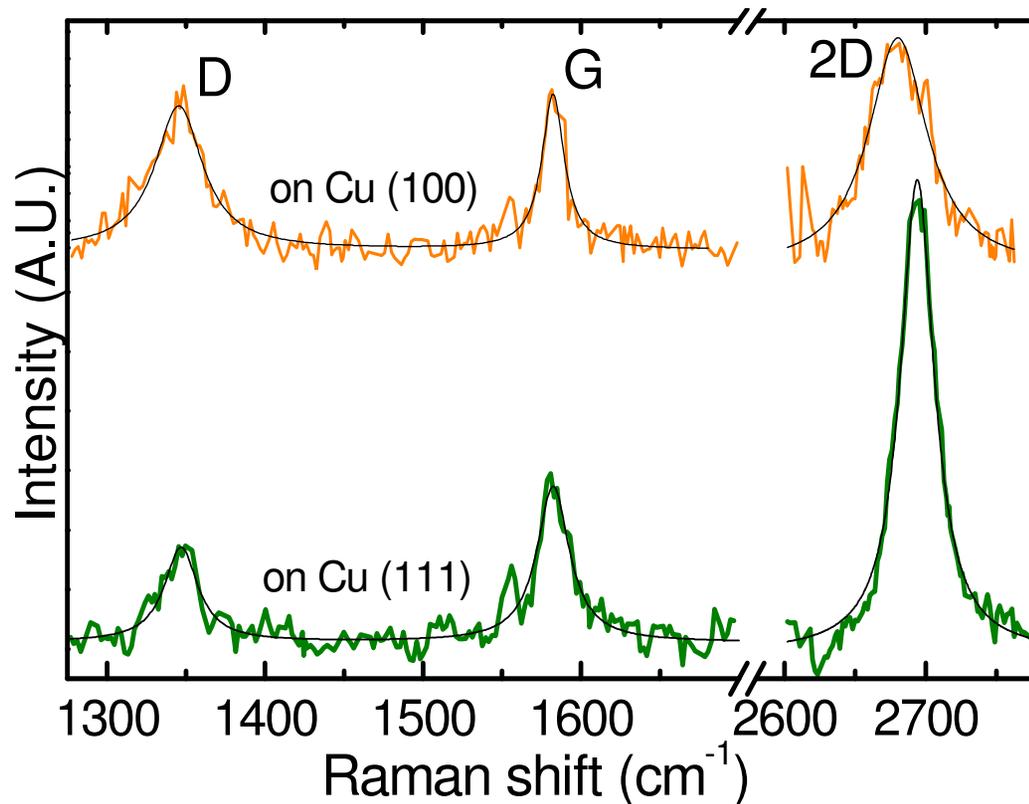
**Blue-shifts indicate compressive physisorption strain**

# CVD graphene on (111) Cu single crystal



- 2D Raman band is **blue-shifted**
- The **blue-shift** indicates compressive strain due to physisorption of graphene on Cu

# Growths on Cu (111) and (100) are vastly different



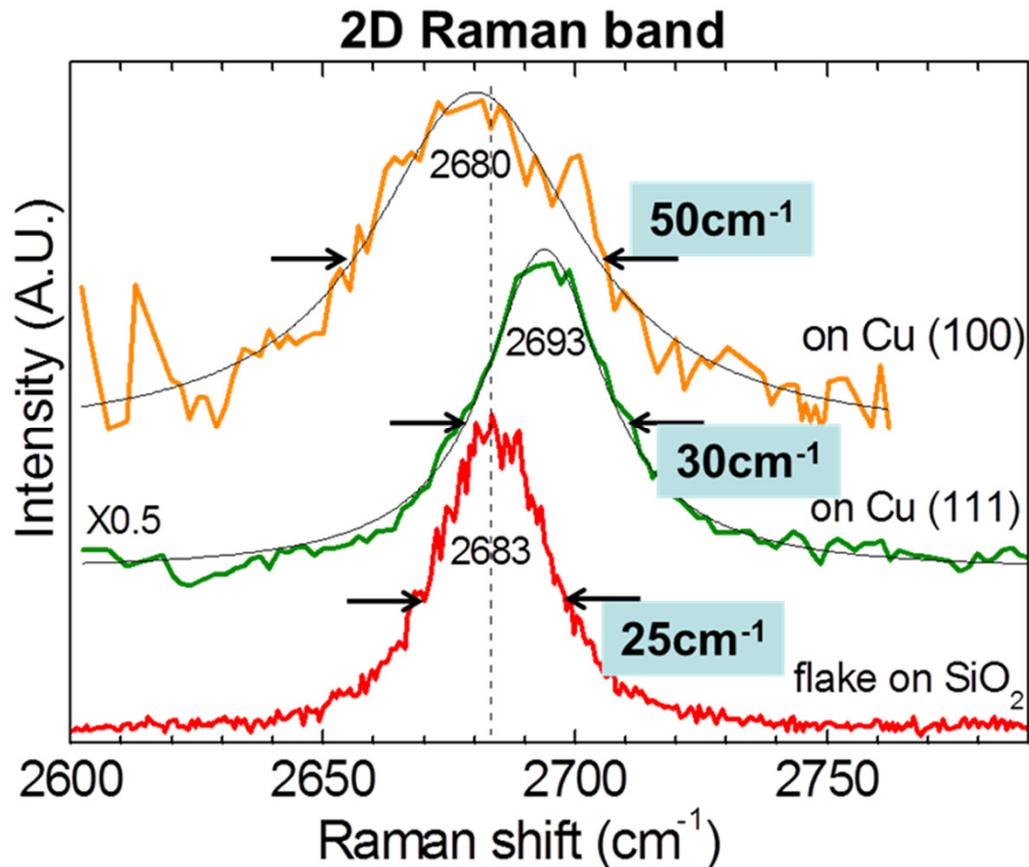
## Graphene on Cu (111):

- Narrower 2D band
- Larger 2D/G ratio
- Smaller D/G ratio

***Graphene on Cu (111) surface has higher quality***

# Graphene on Cu single crystal

## Nonuniform strain

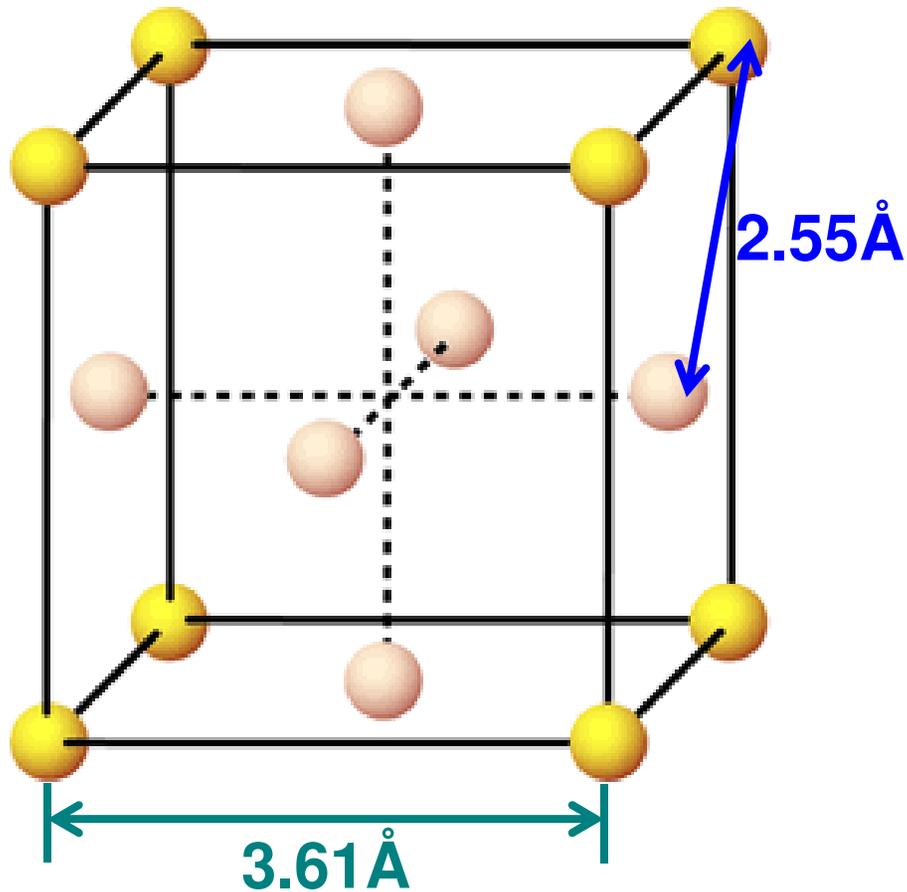


- graphene on Cu (111): 2D band is broadened by 20%
- graphene on Cu (100): 2D band is broadened by a factor of 2

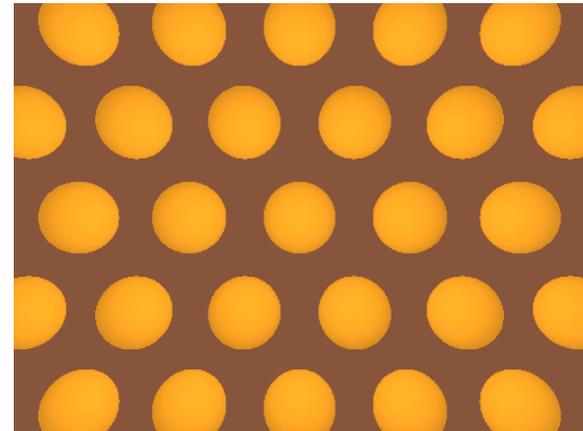
*Strain is more uniform on Cu (111) surface*

# Cu single crystal

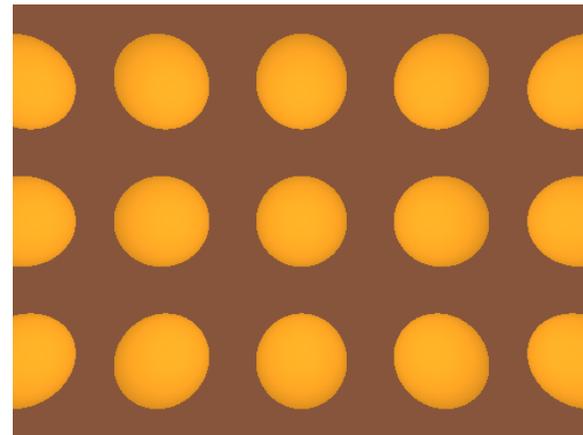
Face-centered cubic  
(FCC) structure



Cu (111) hexagonal



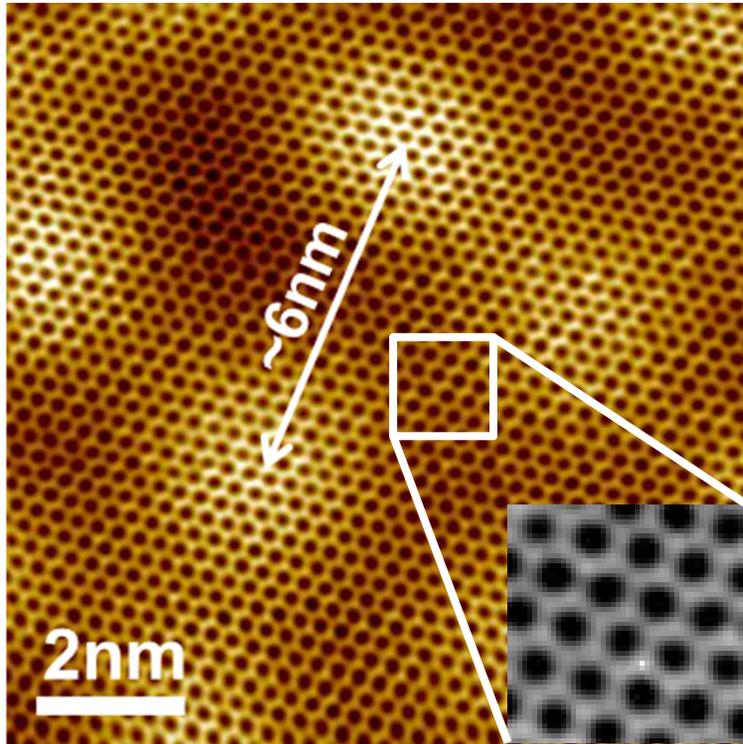
Cu (100) square



# Characterization by scanning tunneling microscopy (STM)

- *Image graphene surface at the atomic level*
- *Moire patterns (superstructure) appear when*
  - \* *two periodic grids with different lattice constants are overlaid*
  - \* *two periodic grids are rotated with each other*

# STM of graphene on Cu (111) single crystal



$$\text{In } k\text{-space } \vec{k}_{\text{Moire}} = \vec{k}_{\text{Cu}} - \vec{k}_{\text{gr}}$$

$$k_{\text{Moire}}^2 = k_{\text{Cu}}^2 + k_{\text{gr}}^2 - 2k_{\text{Moire}}k_{\text{gr}} \cos \theta$$

$$|\vec{k}_{\text{Moire}}| = \frac{1}{6\text{nm}}$$

$$|\vec{k}_{\text{Cu}}| = \frac{1}{0.255\text{nm}}$$

$$|\vec{k}_{\text{gr}}| = \frac{1}{0.246\text{nm}}$$

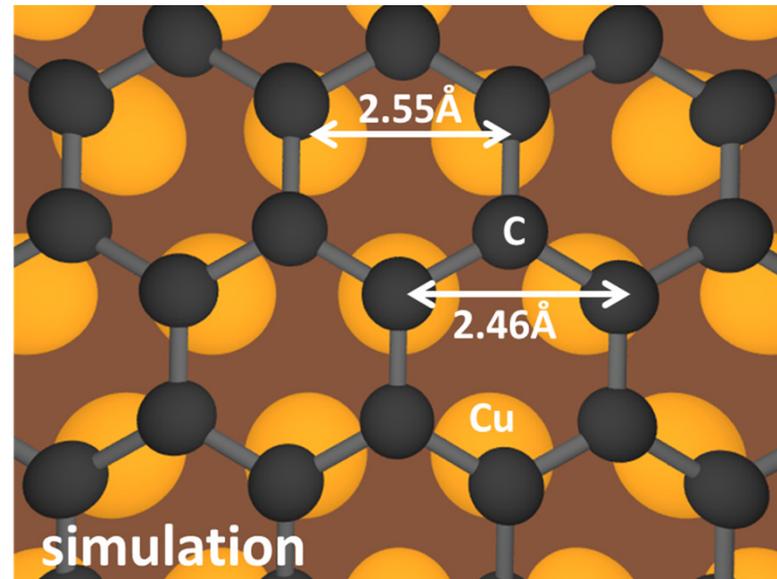
$$\left. \begin{array}{l} |\vec{k}_{\text{Moire}}| = \frac{1}{6\text{nm}} \\ |\vec{k}_{\text{Cu}}| = \frac{1}{0.255\text{nm}} \\ |\vec{k}_{\text{gr}}| = \frac{1}{0.246\text{nm}} \end{array} \right\} \rightarrow \theta \approx 0^\circ$$

Rotational angle  $\theta$  is almost zero

# Graphene on Cu (111) single crystal

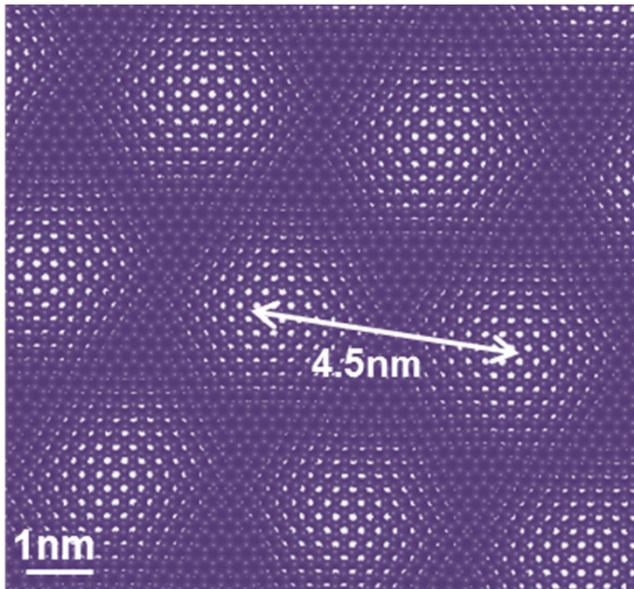
- Graphene **honeycomb** lattice with lattice constant of **2.46Å**
- Cu (111) **hexagonal** lattice with lattice constant of **2.55Å**
- **Rotational angle  $\theta$  is almost zero**

Quasi-epitaxial  
growth

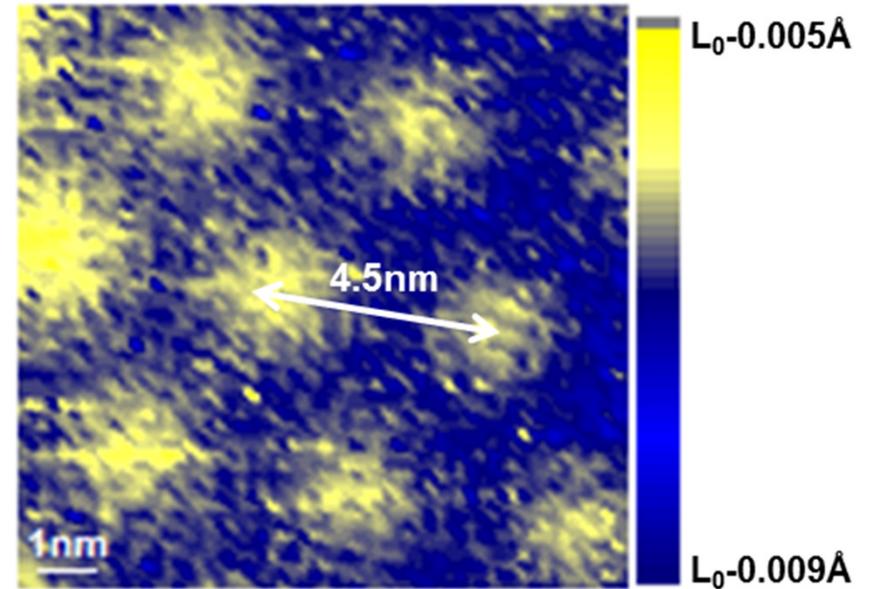


# Molecular dynamics (MD) simulations of graphene on Cu (111) single crystal

Simulated superstructure



C-C bond length simulation

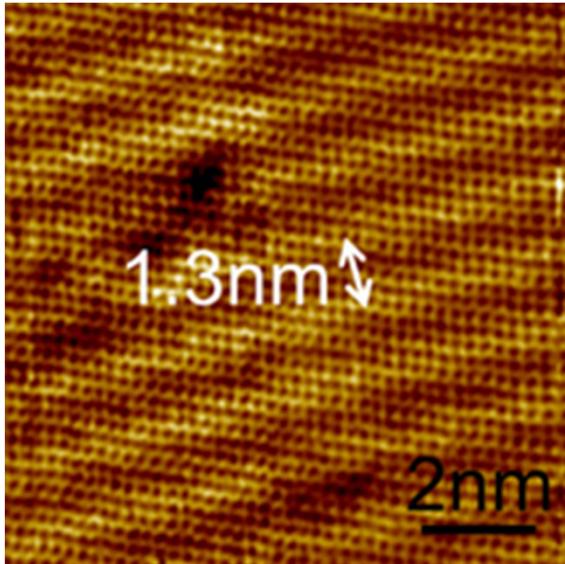


- Compressive strain over the entire surface
- Strain varies between 0.3% and 0.6%
- Domain walls are highly compressed

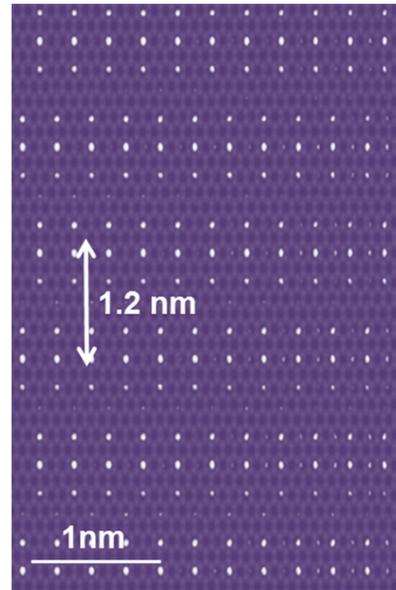
# Graphene on Cu (100) single crystal

*Molecular Dynamics simulations*

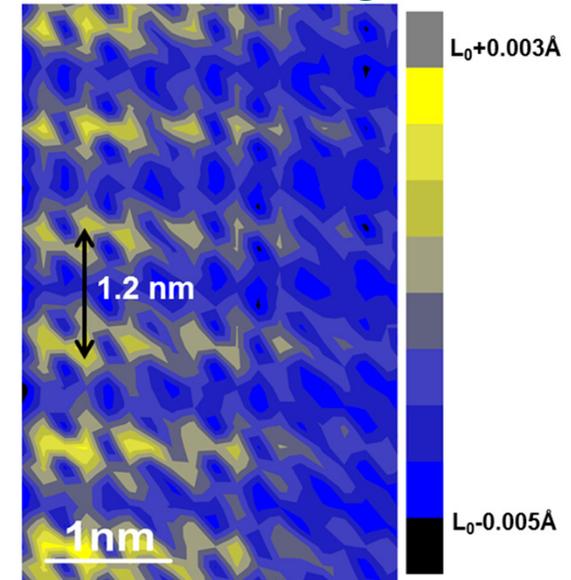
STM



Superstructure



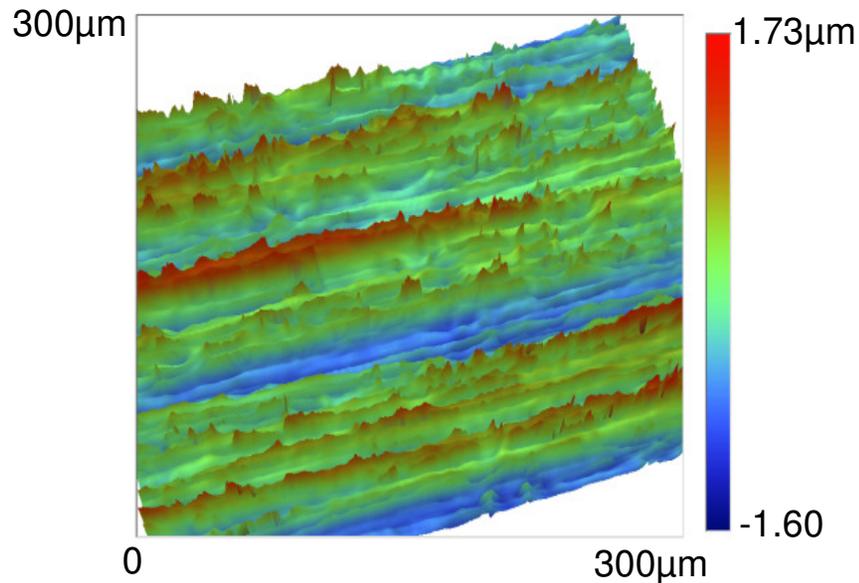
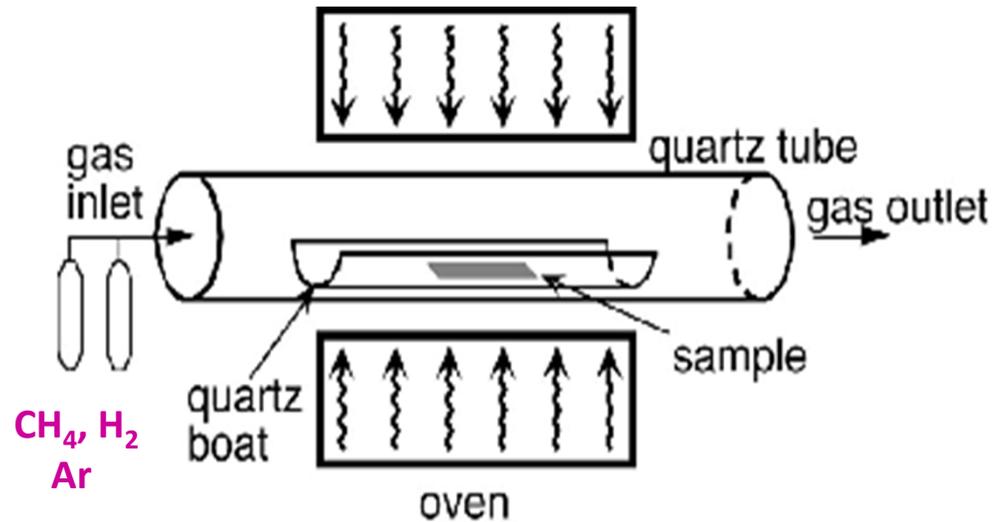
C-C bond length



- **No epitaxy:** honeycomb graphene lattice on Cu (100) square lattice
- Highly nonuniform strain that varies from 0.3% compressive to 0.2% tensile

# CVD growth on Cu foil

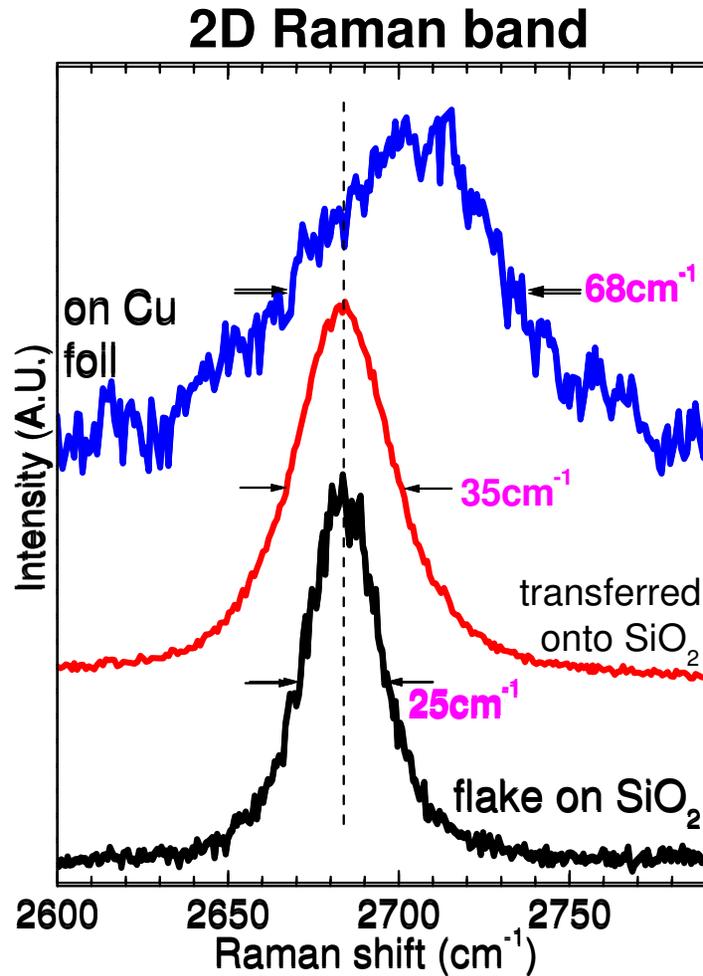
- Growth temperature 1000°C
- Methane was used as C source



*polycrystalline and rough surface*

# CVD graphene on Cu foil

## Release of strain



after transfer onto  $\text{SiO}_2$ :

- *2D band overlaps that of natural graphene*
- *broadening related to residual strain*

# CVD graphene on copper substrates

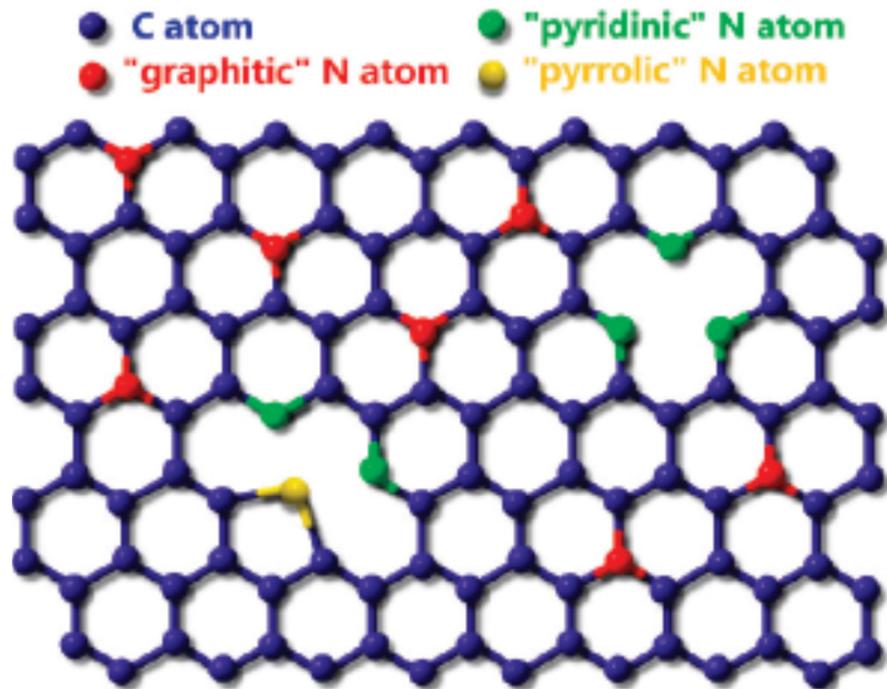
- Raman and STM reveal degree of perfection and strain
- Strain is dependent on Cu surface orientation
- Release of strain after graphene is transferred onto SiO<sub>2</sub> substrate

*Nano Letters* 12, 2408 (2012)

# Graphene

- Introduction
- Graphene on Cu single crystal surfaces
- **Nitrogen doped graphene**
- Twisted bilayer graphene

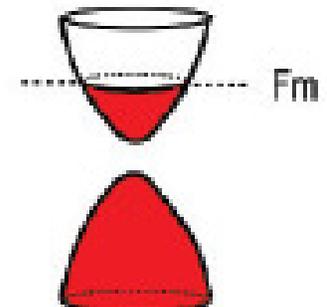
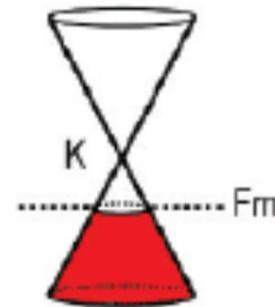
# Nitrogen doped graphene



**Doping:** control electronic properties of graphene

pristine graphene

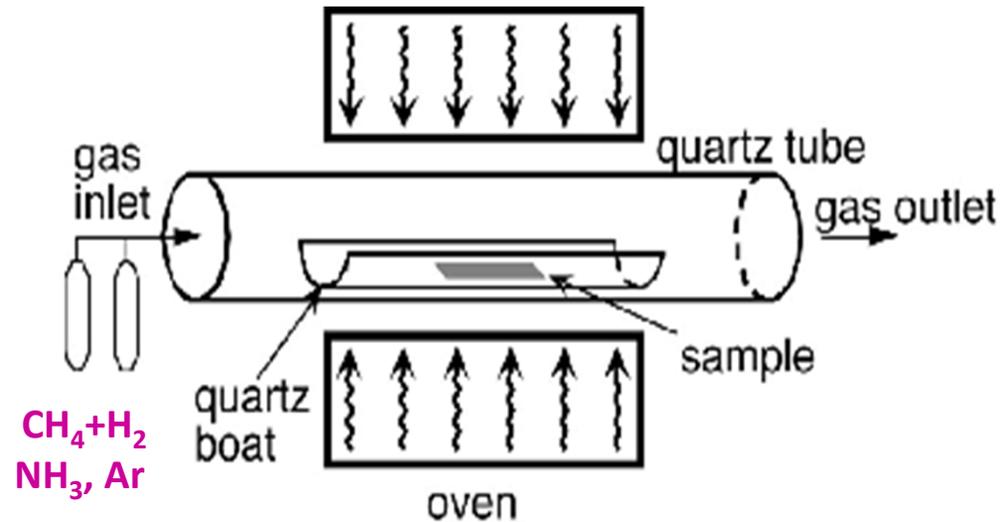
N-doped graphene



Wei et al. *Nano Lett.* **9**, 1752 (2009)

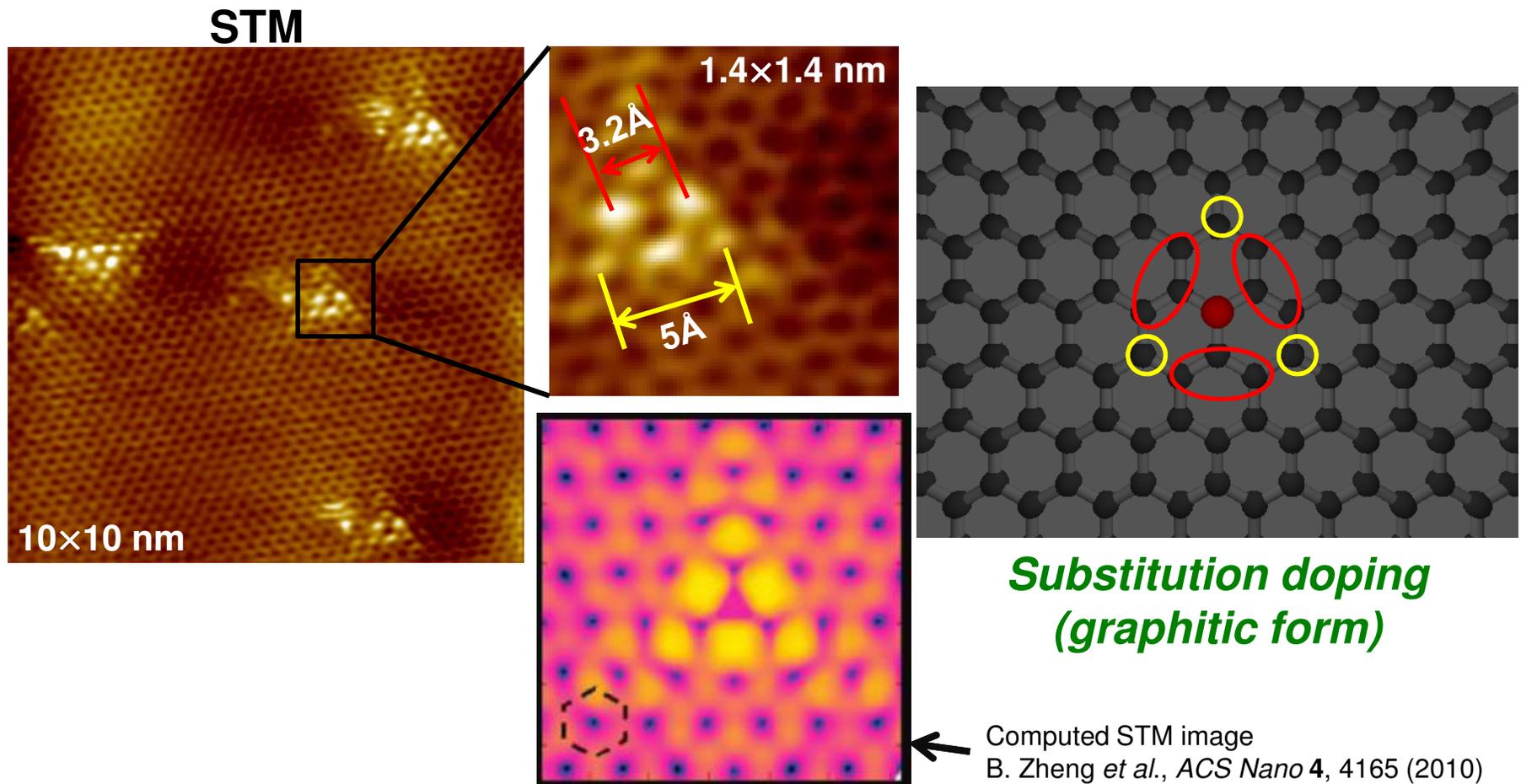
# Nitrogen doped graphene

## *chemical vapor deposition (CVD)*

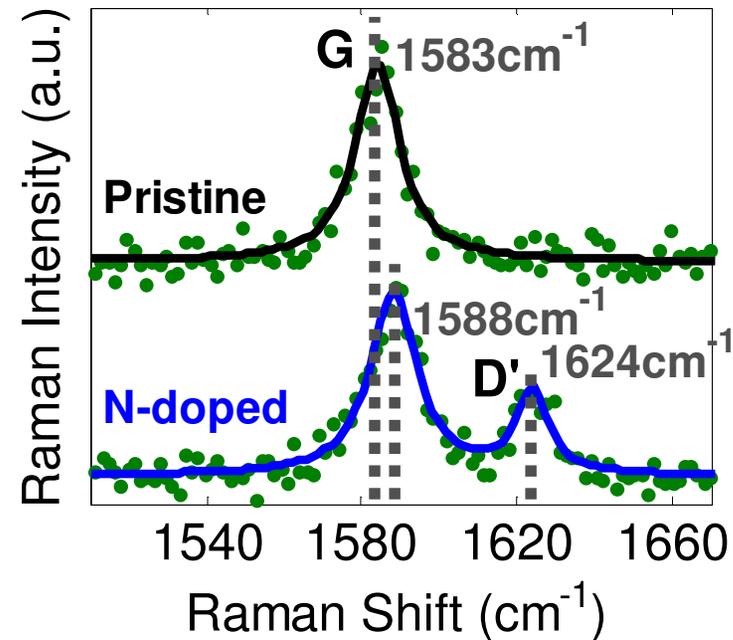
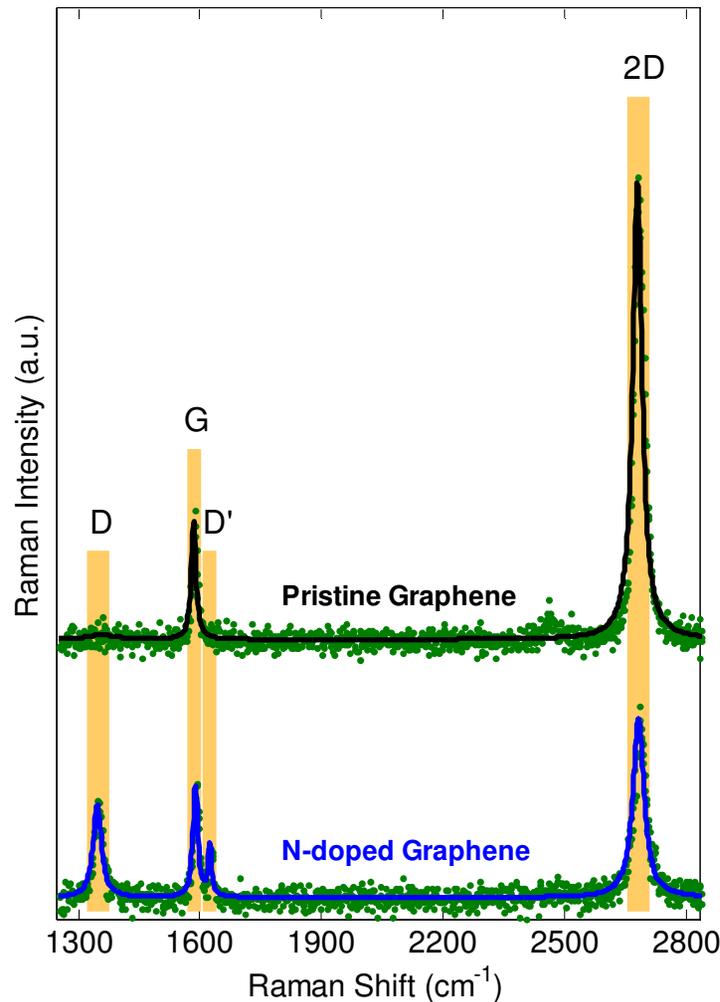


- Copper foil is heated to  $\sim 1000^\circ\text{C}$
- $\text{NH}_3$ ,  $\text{CH}_4$ ,  $\text{H}_2$ , and argon gas are passed in with appropriate ratio
- Doping level is controlled by the pressure of ammonia gas

# Visualizing individual nitrogen dopants in honeycomb lattice



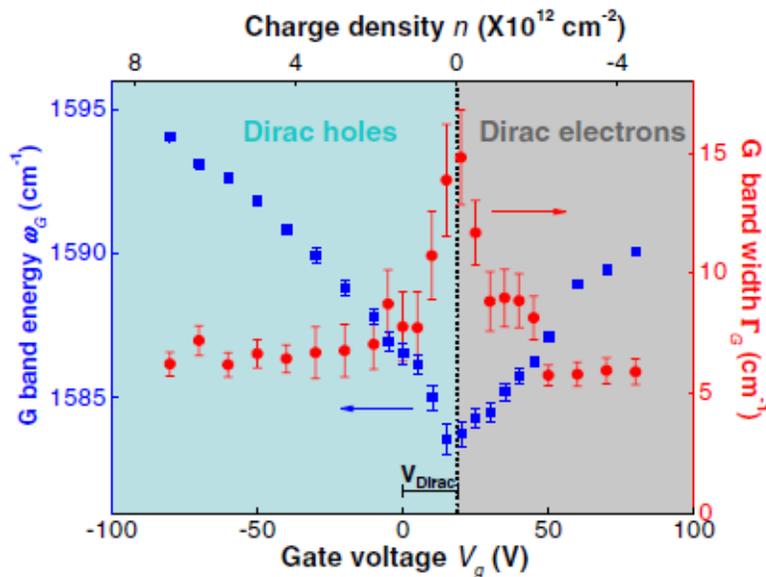
# Raman scattering from nitrogen doped graphene



- strong defect bands D and D'
- lower 2D/G ratio
- blueshift of G peak

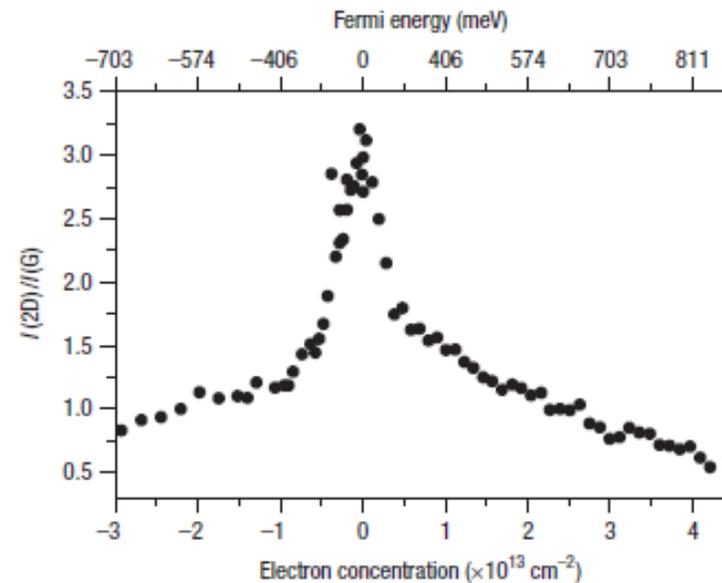
# Raman characterization of carrier density in graphene

## G peak position



Yan *et al.*, *Physical Review Letters*  
98, 166802 (2007)

## 2D/G ratio

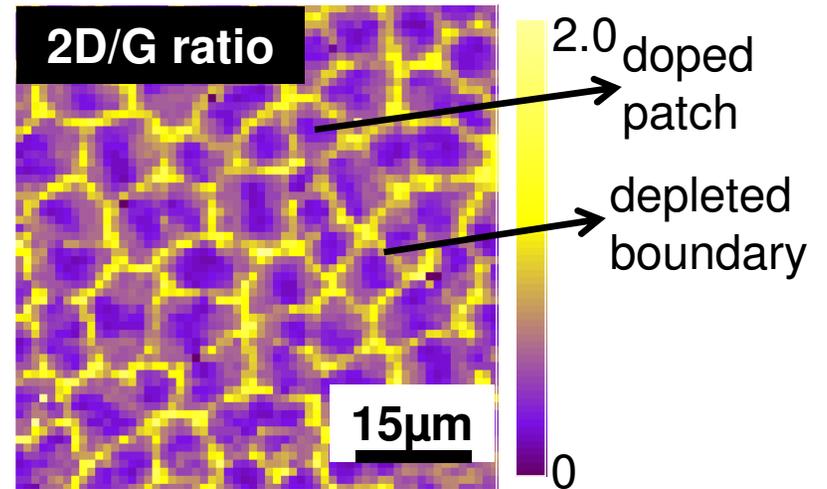
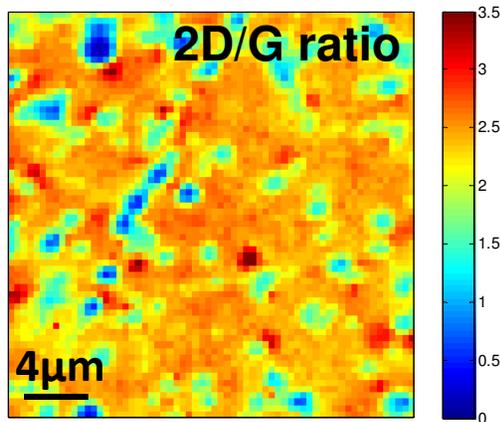


Das *et al.*, *Nature Nanotechnology*  
3, 210 (2008)

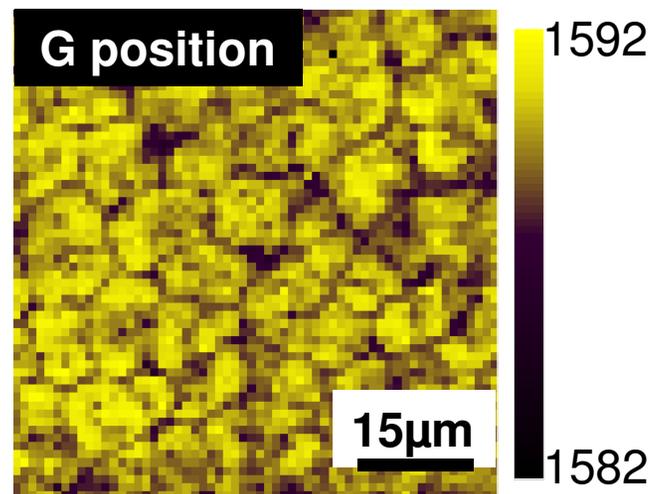
**lower electron (hole) density  $\longleftrightarrow$   
lower G peak frequency and higher 2D/G ratio**

# Raman mapping of nitrogen doped graphene

pristine graphene

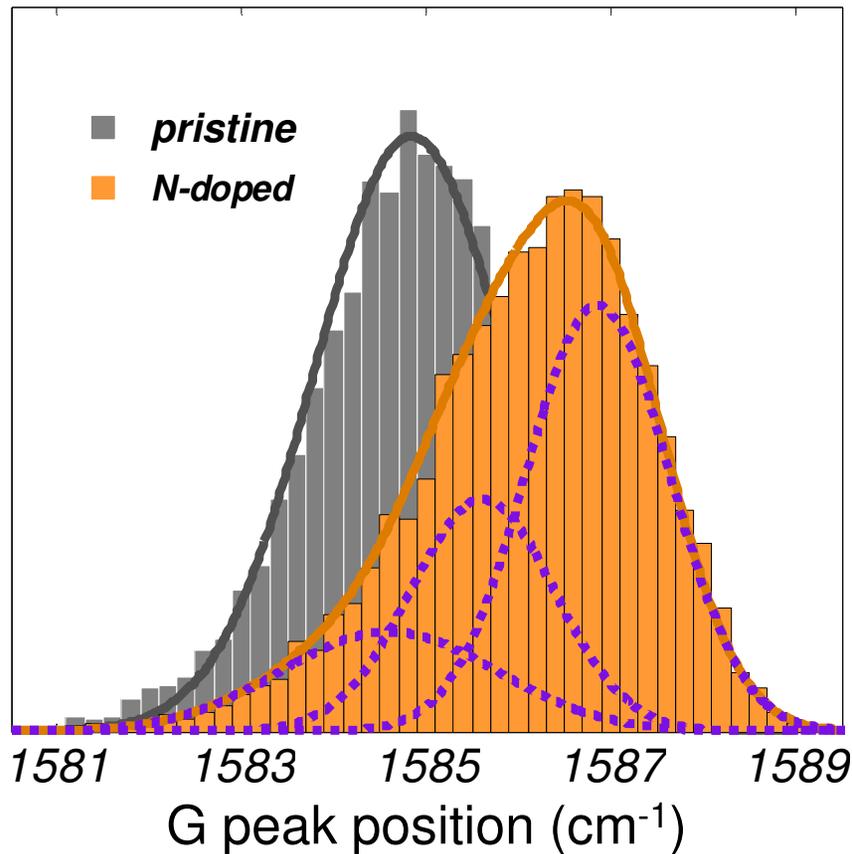


Lower dopant (nitrogen) concentration at the grain boundaries



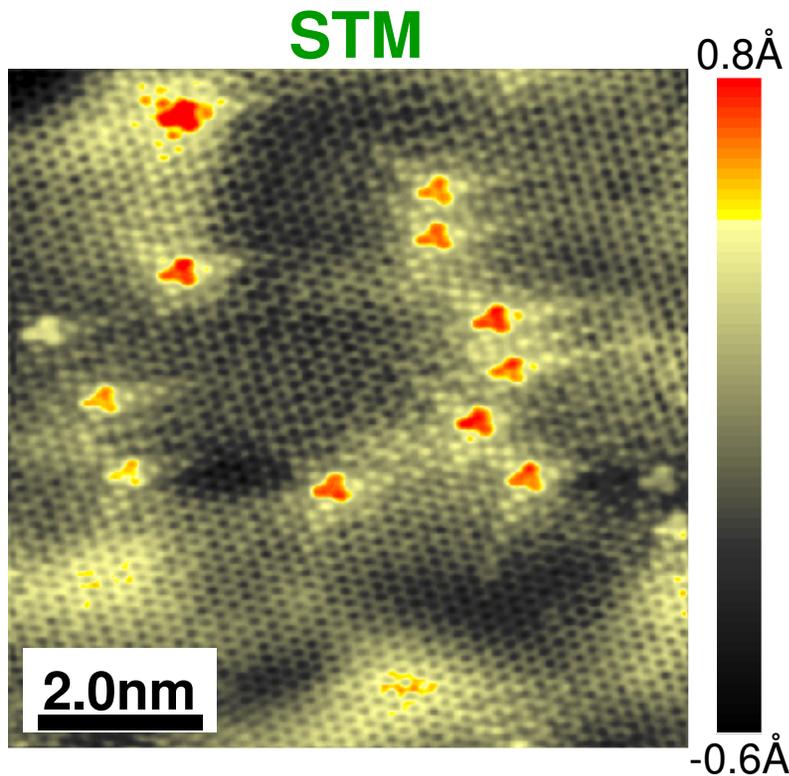
# Estimation of average carrier concentration

## Distribution of G peak position



Based on the shift in the position of the G peak in the Raman spectra induced by doping, we estimate that the carrier concentration is  $5 \pm 1.5 \times 10^{12} \text{ cm}^{-2}$

# Mobile carriers contributed by nitrogen dopants



- Nitrogen dopants are randomly distributed in the honeycomb lattice
- N atom density  $\sim 1.3 \times 10^{13} \text{ cm}^{-2}$
- Carrier concentration (estimated by Raman) is  $5 \pm 1.5 \times 10^{12} \text{ cm}^{-2}$

*Each graphitic N dopant contributes (on average)  $\sim 0.4$  mobile carriers to the graphene lattice*

# Nitrogen doped graphene

- Individual nitrogen dopant was observed in real space by STM. Most of the dopants are graphitic forms
- Nitrogen concentration is lower at grain boundaries
- Each graphitic nitrogen atom contributes  $\sim 0.4$  mobile charge carrier (electron) into graphene lattice

***Science 333, 999 (2011)***

# Outline

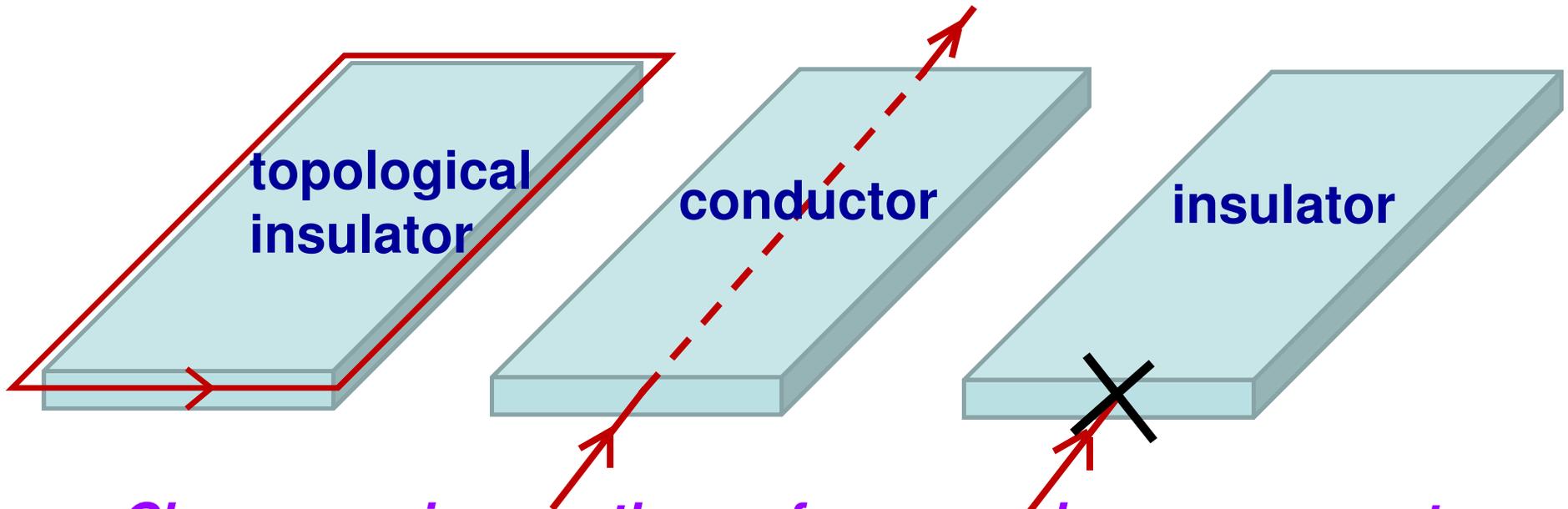
## *Graphene*

- Introduction
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- Twisted bilayer graphene

## ***Topological insulator nanostructures***

- **Bi<sub>2</sub>Te<sub>3</sub> nanoplates**

# Topological insulator (TI)



*Charge carriers on the surfaces or edges carry a net spin and conduct electricity without much thermal dissipation*

→ promising candidate for spintronics and quantum computing devices which are based on spin transport

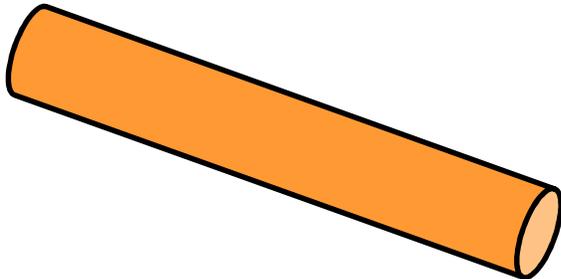
# Challenges in TI research



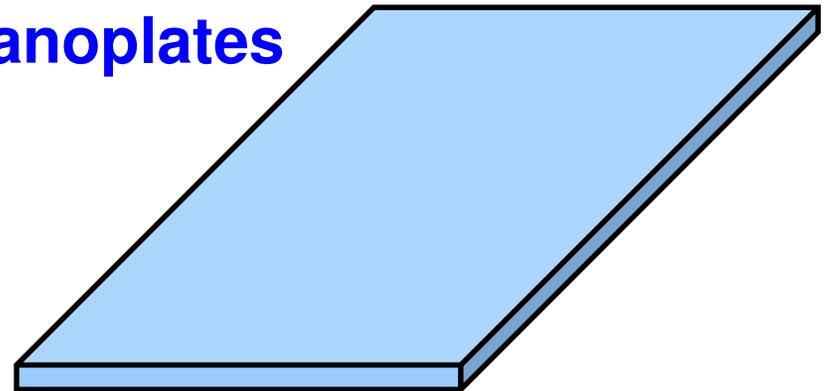
In 3D TIs, bulk characteristics dominate their properties

Reduce its dimension:  
enhance surface-to-bulk ratio

1D nanowires



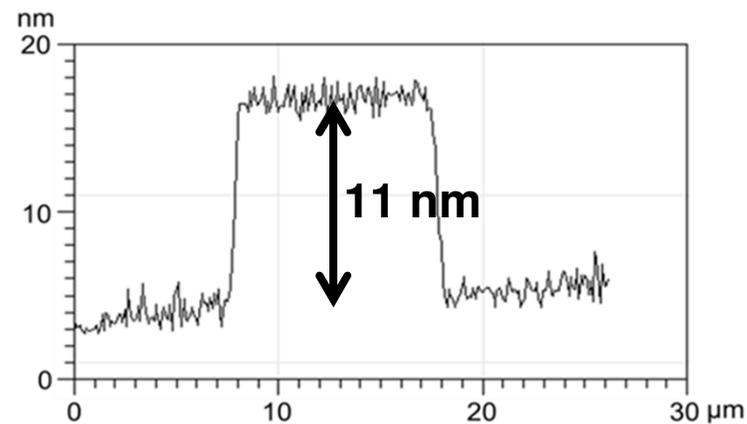
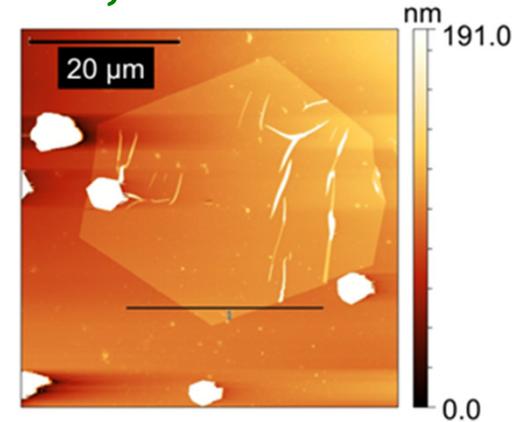
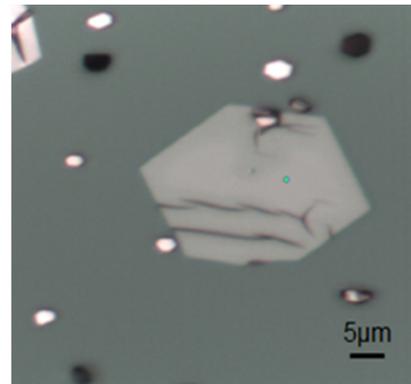
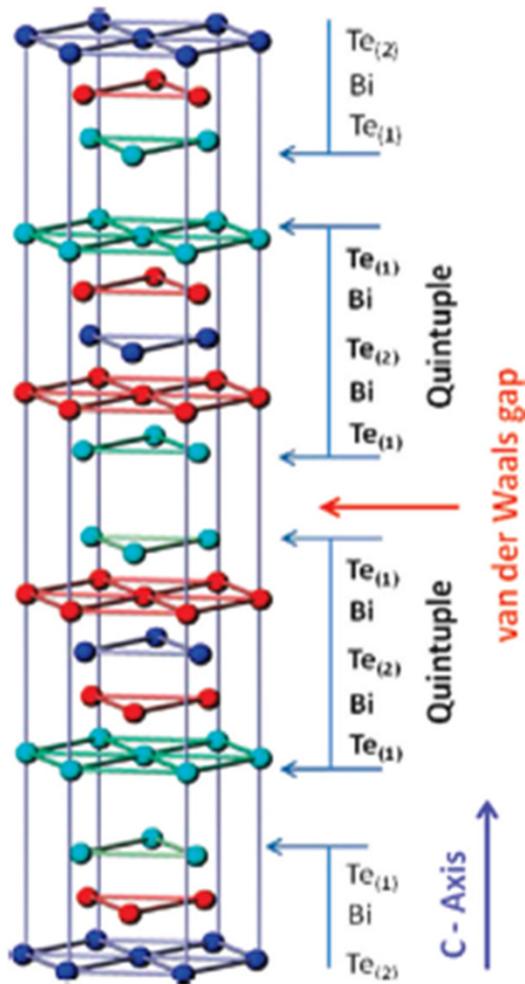
2D nanoplates



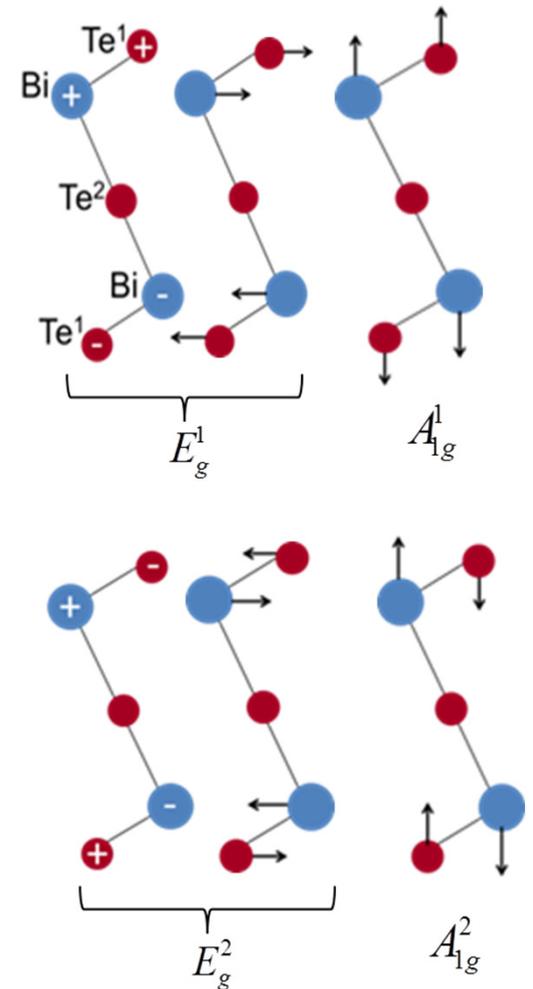
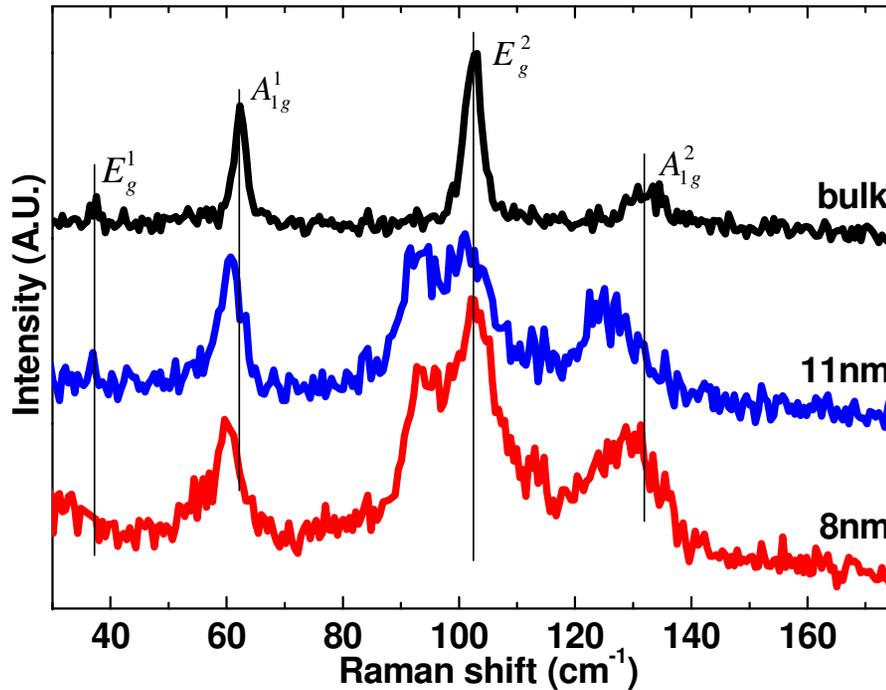
# TI nanoplates

representative TI material:  $\text{Bi}_2\text{Te}_3$

As-grown NP, 11 nm



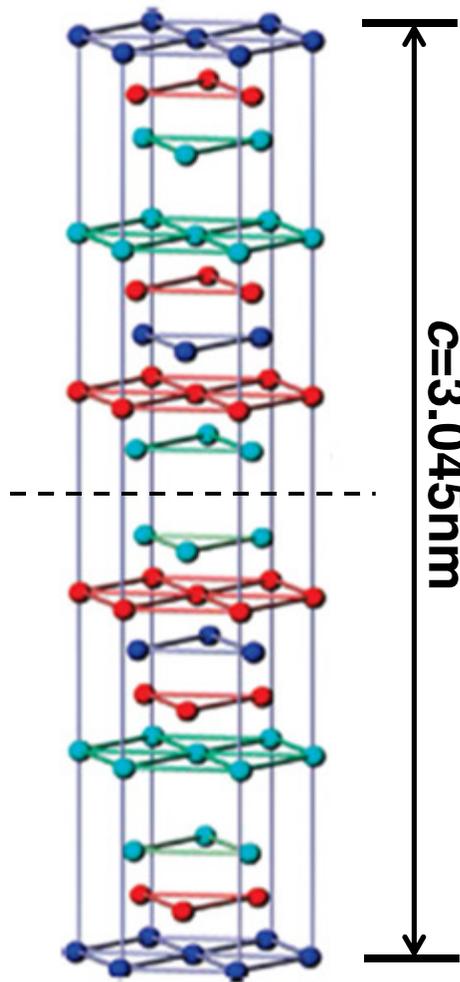
# Raman scattering from bulk $\text{Bi}_2\text{Te}_3$ and as-grown $\text{Bi}_2\text{Te}_3$ NPs



**4 Raman-active modes from bulk  $\text{Bi}_2\text{Te}_3$**

The out-of-plane modes are more sensitive to the reduction of thickness

# Raman selection rules in crystals with inversion symmetry



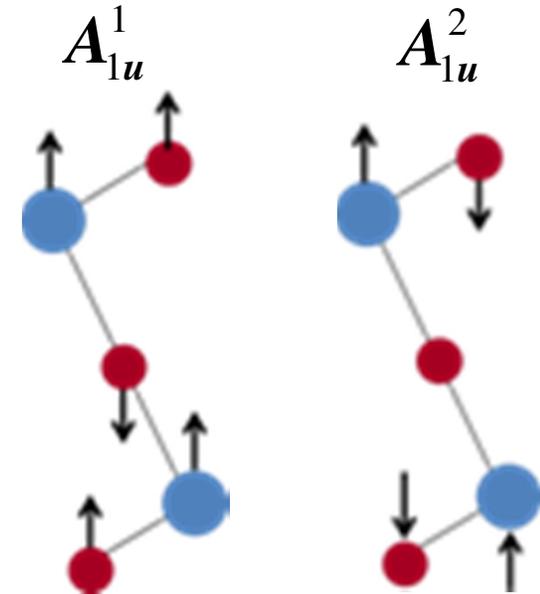
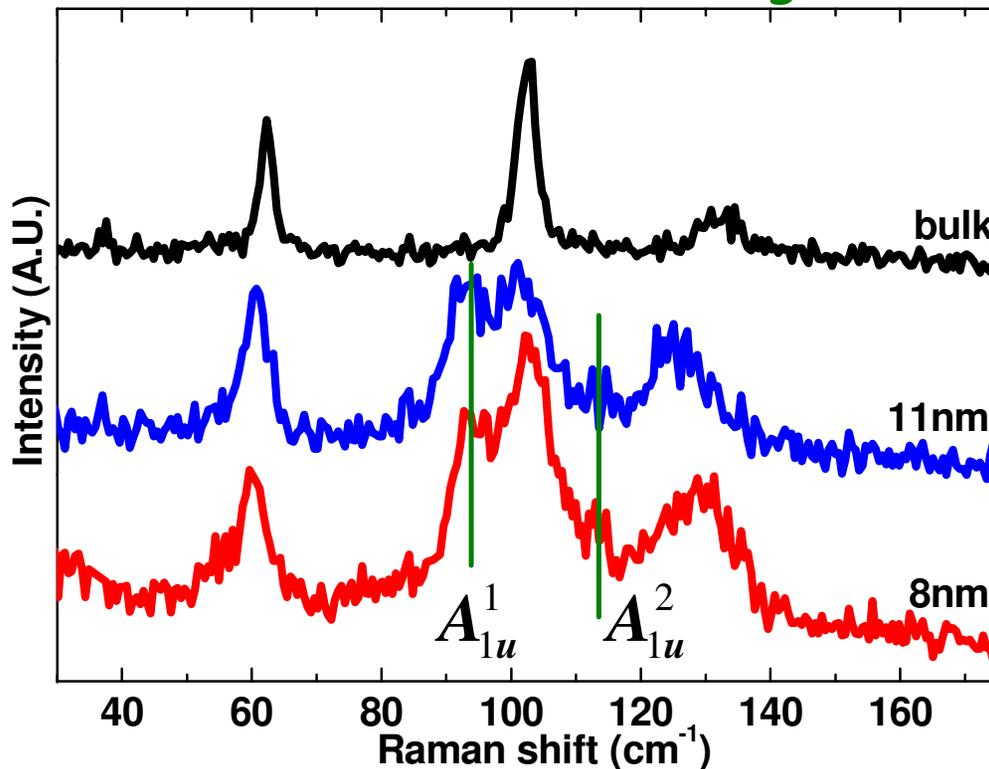
**Bi<sub>2</sub>Te<sub>3</sub> crystal lattice**

**Even-parity** phonons (with subscripts “*g*”) can be observed in **Raman scattering**

**Odd-parity** phonons (with subscript “*u*”) can be observed in **infrared absorption**

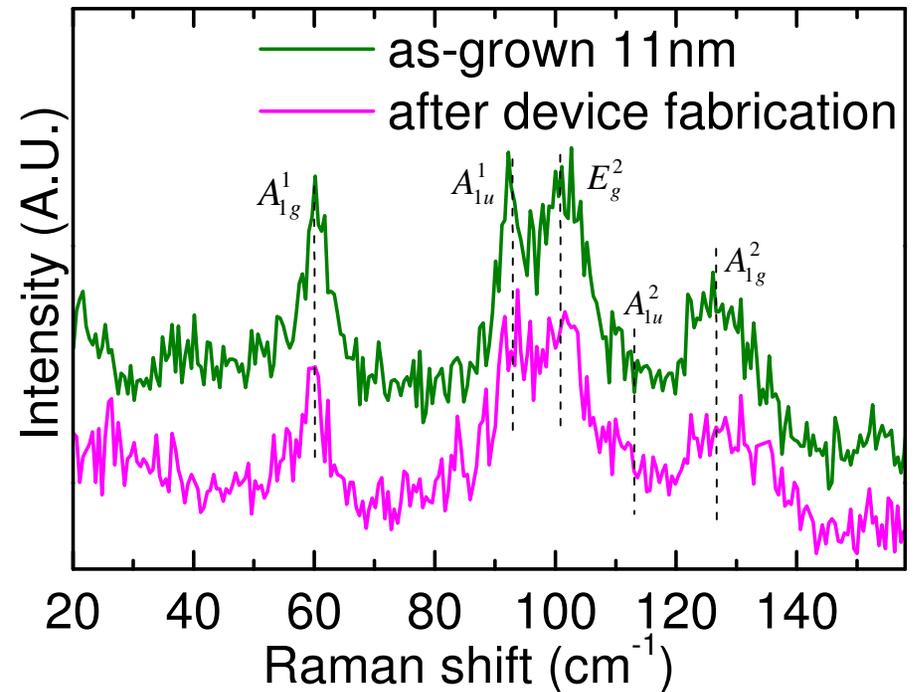
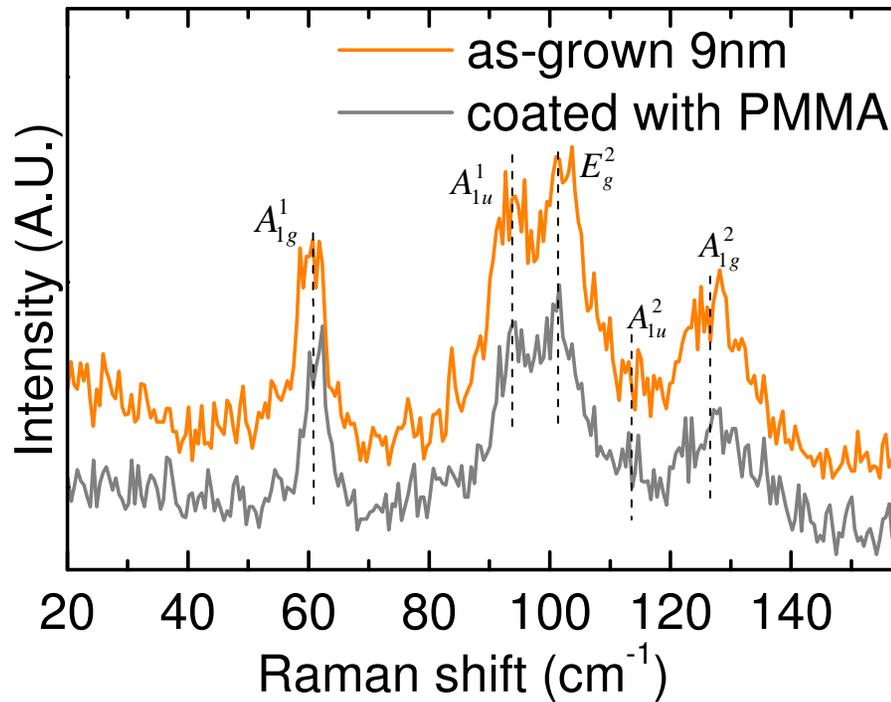
# Infrared-active modes observed in Raman scattering from $\text{Bi}_2\text{Te}_3$ NPs

Two new modes emerge



The emergence of *IR-active* modes in *Raman spectra* reveals a *breakdown of inversion symmetry* in TI nanostructures

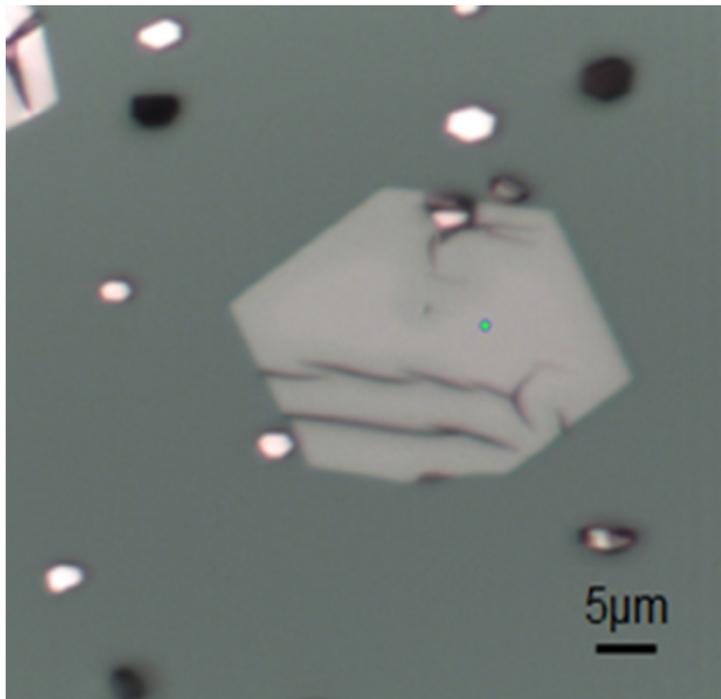
# Robustness of vibrational properties of $\text{Bi}_2\text{Te}_3$ NPs



**Vibrational properties preserve  
after device fabrication**

# Transferred $\text{Bi}_2\text{Te}_3$ NPs

As-grown 11 nm

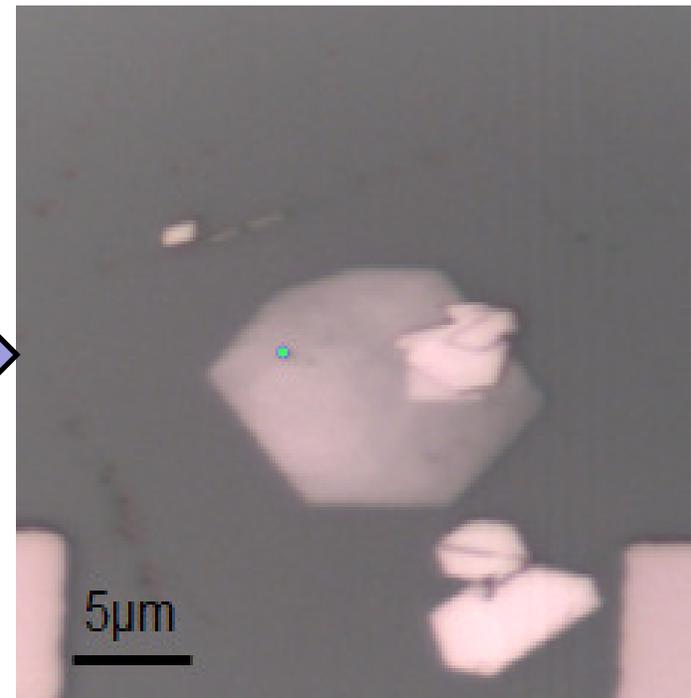


Mechanical  
transfer to



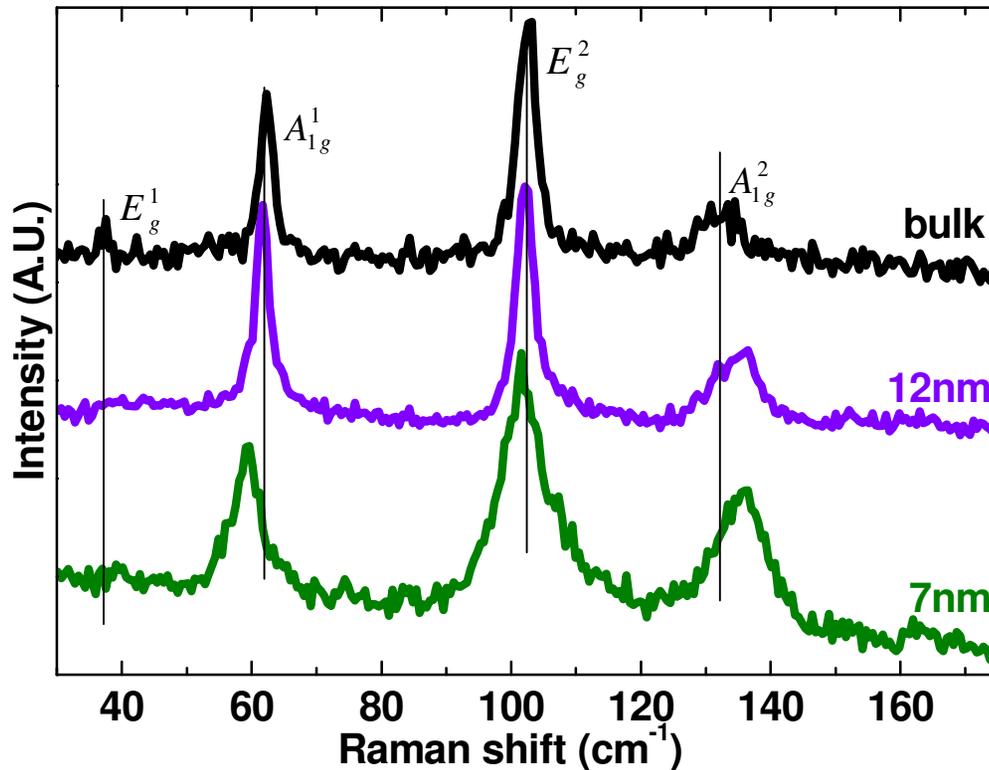
remove  
impurities

Transferred 12 nm

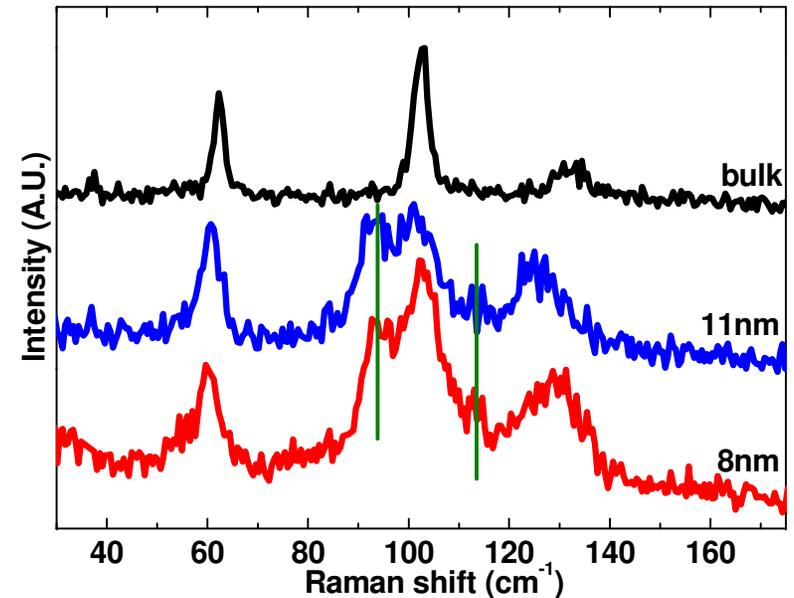


# Raman scattering from transferred $\text{Bi}_2\text{Te}_3$ NPs

## Transferred NPs



## As-grown NPs



***Infrared-active modes are absent in transferred NPs:  
absence of inversion symmetry breaking***

# Absence of inversion symmetry breaking in transferred NPs

***Possible reason:***

**The transferred NPs are the ones that stand on edge on the substrate**

***strain applied by the substrate induce inversion symmetry breaking in as-grown NPs***

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